

Lubrication Effects on Surface Pitting and Scuffing in Gears - A Review

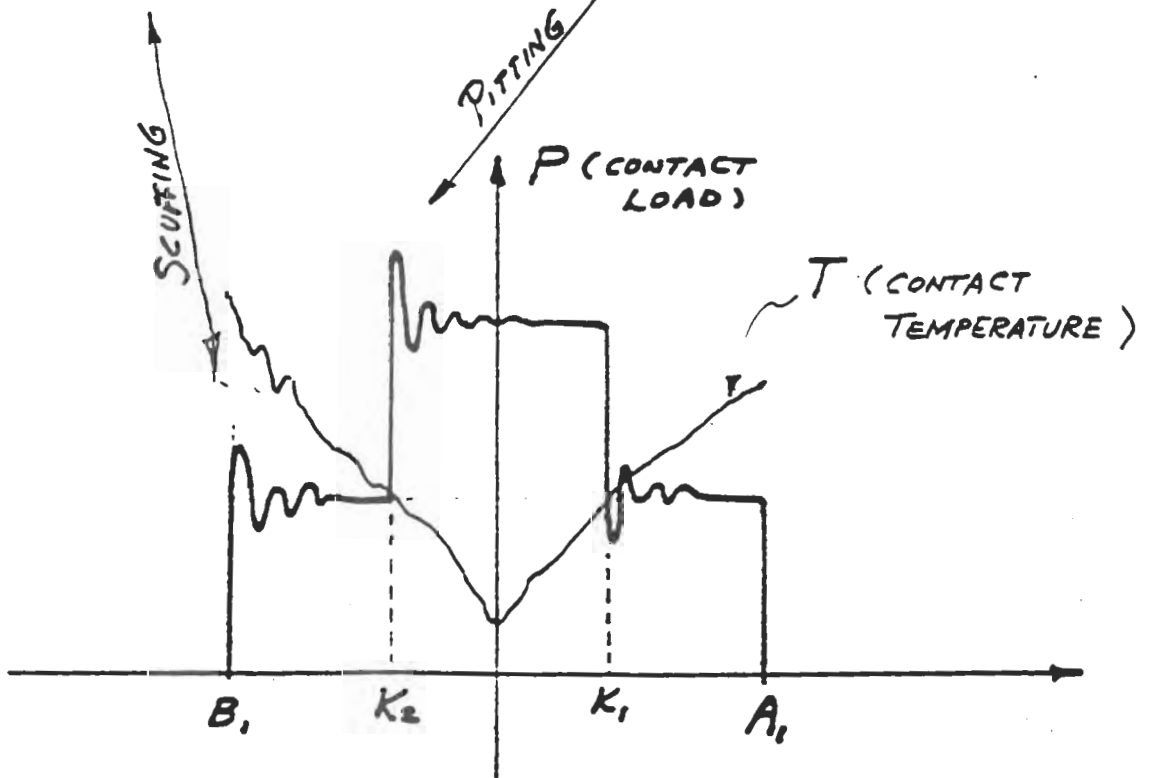
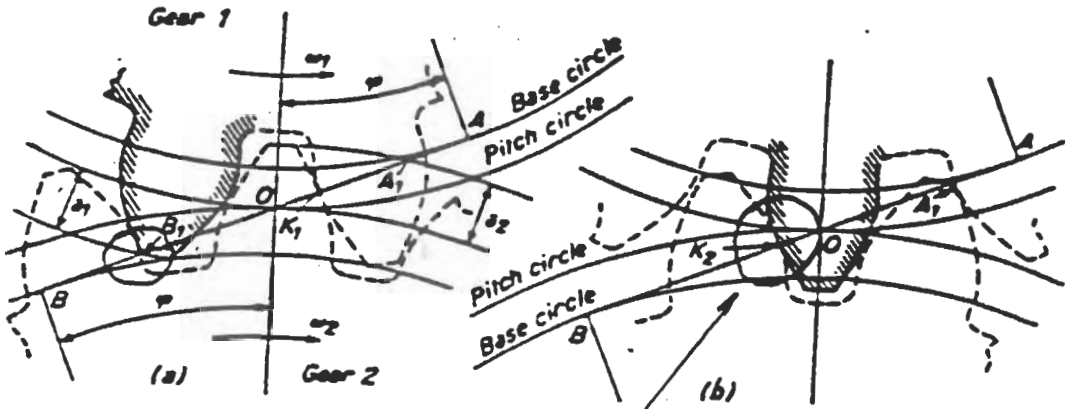
H. S. Cheng

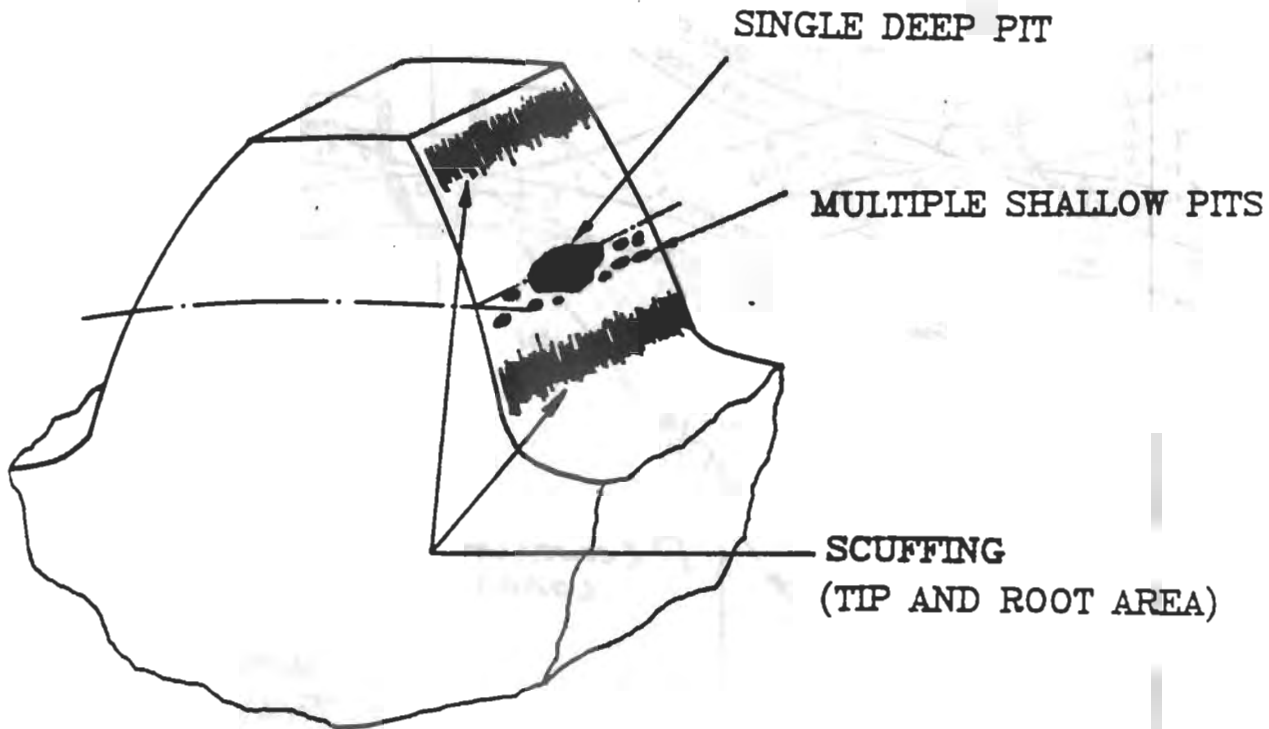
Center for Engineering Tribology
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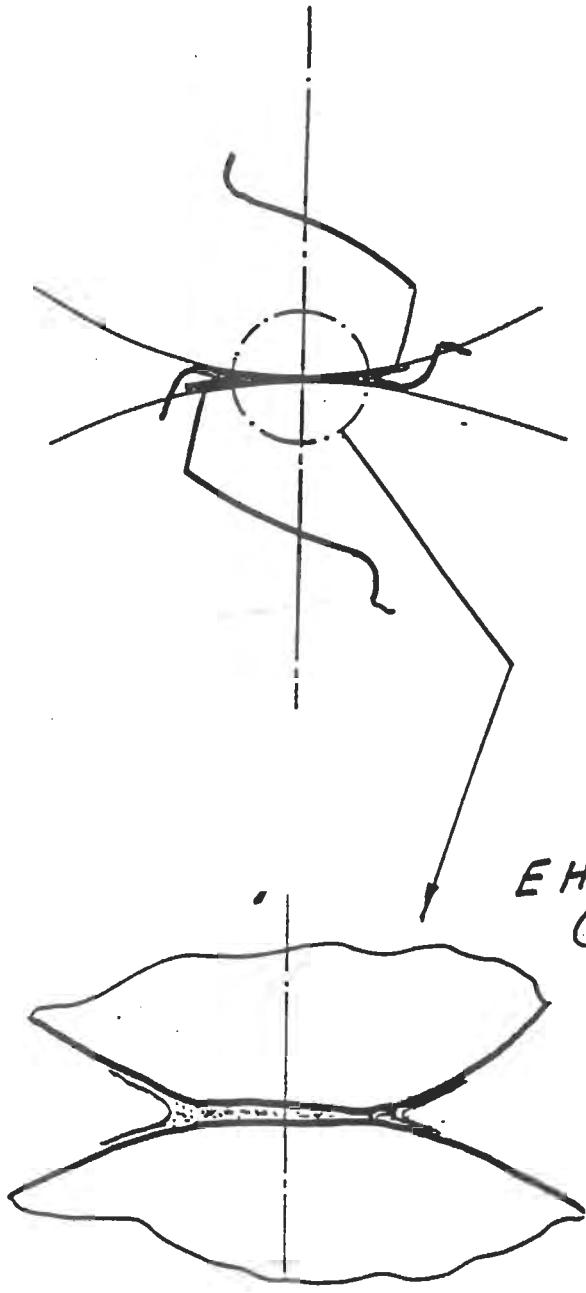
Evanston, IL 60208

Outline of Gear Lubrication Review

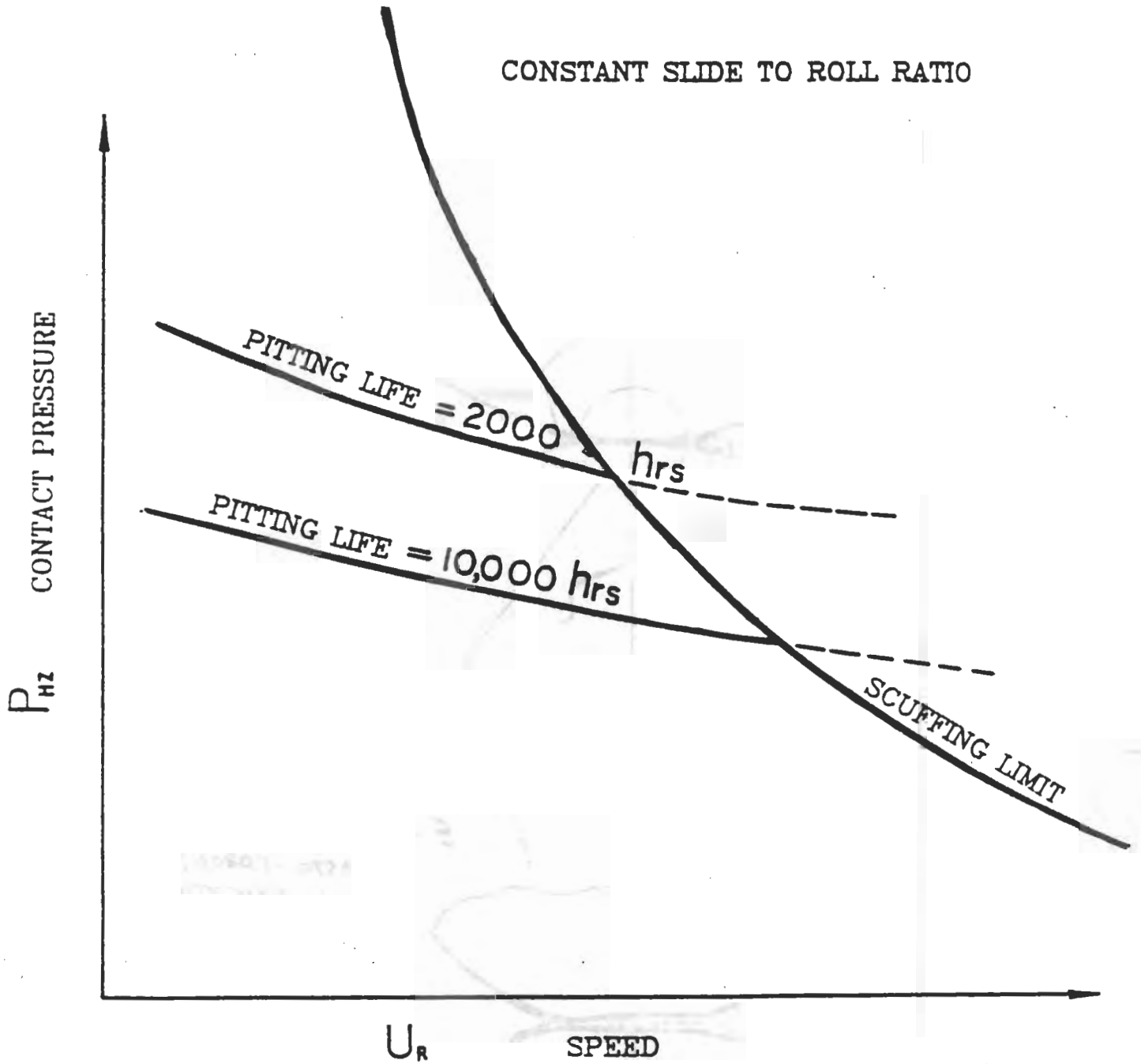
- Tribofailure Modes in Gears
- Lubrication Performance
- Mixed Lubrication Characteristics
 - \bar{h} and h^* - Average and Asperity Film Thickness
 - \bar{p} and p^* - Average and Asperity Pressure
 - \bar{T} and T^* - Average and Asperity Contact Pressure
- Surface Fatigue Phenomenon
- Crack Initiation Modeling
- Crack Propagation Modeling
- Gear Teeth Contact Fatigue Life Modeling
- Scuffing Phenomenon
- Roller Simulation and Scuffing Experiments
- Scuffing Modeling
- Concluding Remarks







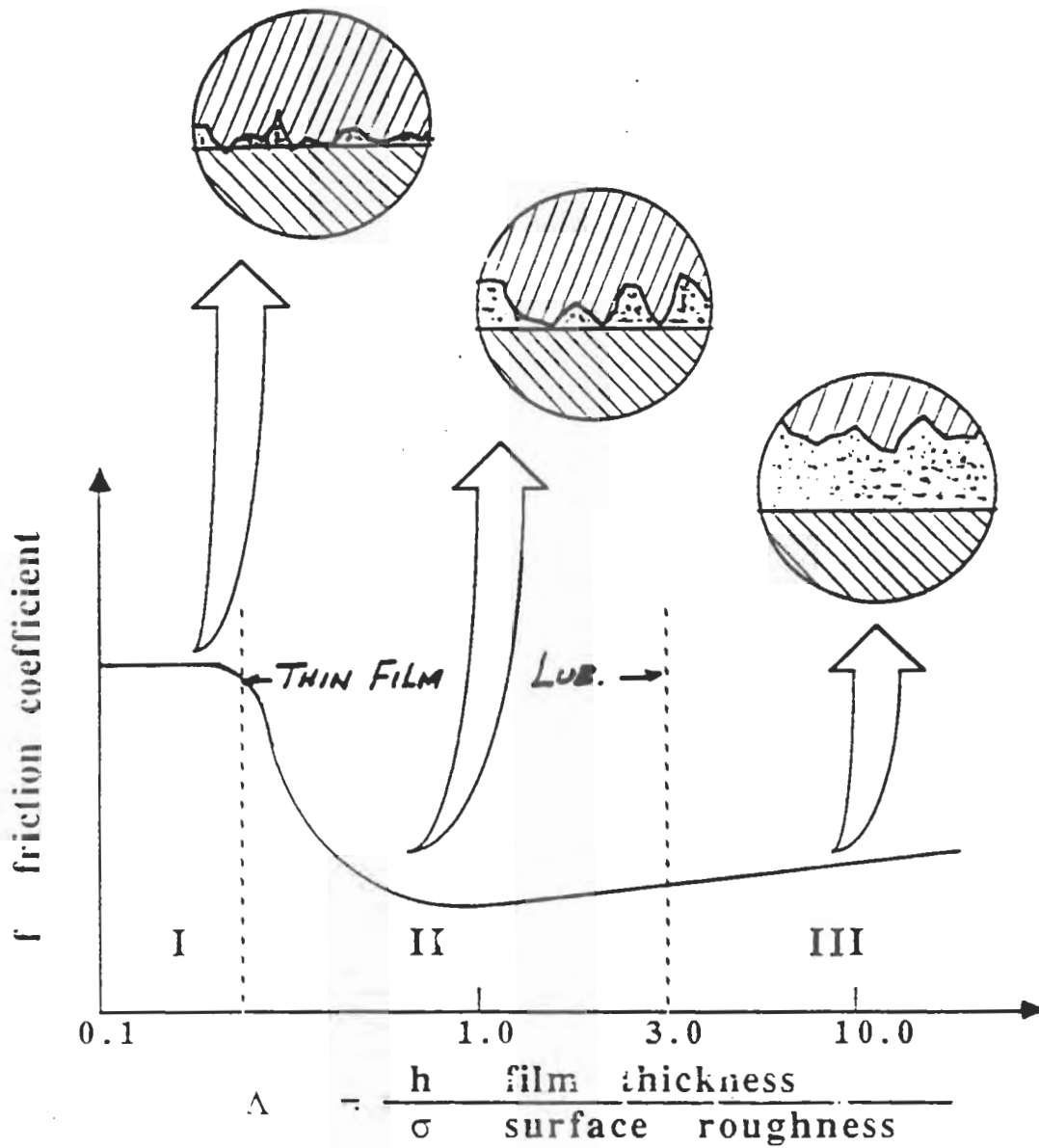
EHD
(ELASTO HYDRODYNAMIC
LUBRICATION.)



SIGNIFICANT EFFECTS :

ROLLING SPEED
 VISCOSITY
 ROUGHNESS
 HARDNESS

INCLUSION
 FIBER ORIENTATION



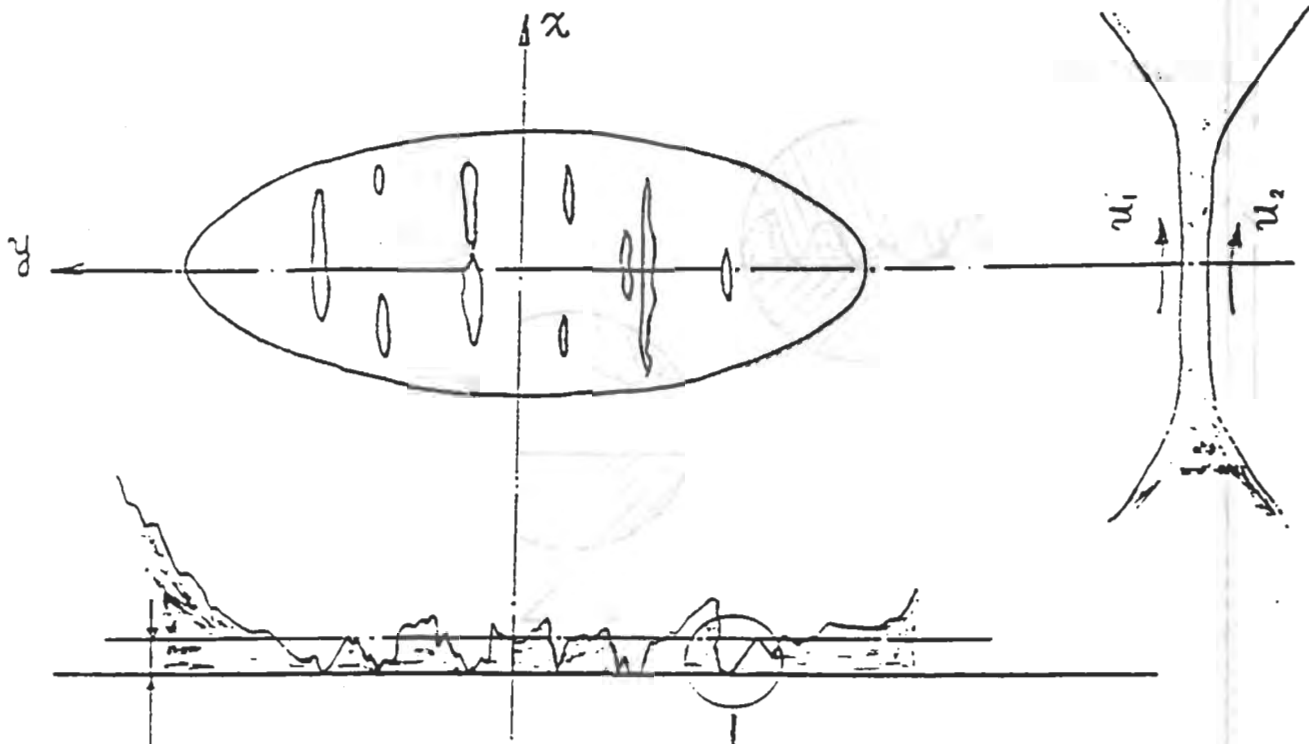
I. Boundary Lubrication

II. Partial E.H.L. (*THIN-FILM LUB.*)

III. Full Film E.H.L.

Regimes of Lubrication in Lubricated Contacts

Mixed-Film Lubricated Contacts



\bar{h} - Average Film

h^* - Micro-EHL Film

\bar{p} , \bar{p}_a - Average Fluid and Contact Pressure

p^* - Asperity Contact Pressure

$\bar{\tau}$, $\bar{\tau}_a$ - Average Fluid and Contact Shear Stress

τ^* - Asperity Contact Shear Stress

\bar{T} , \bar{T}_a - Average Fluid and Contact Surface Temperatures

T^* - Asperity Contact Temperature

CURVE 1 - STANDARD
 CURVE 2 - NON-STANDARD

CURVE No.	1	2	CURVE No.	1	2
NUMBER OF TEETH (P)	28	28	MODULE	5.0	5.0 (MM)
NUMBER OF TEETH (G)	33	33	PRESSURE ANGLE	20.0	25.0 (DEG)
ADDENDUM OF PINION	5.00	6.50 (MM)	CLEARANCE	1.25	1.25 (MM)
ADDENDUM OF GEAR	5.00	3.50 (MM)	FILLET RADIUS	1.50	1.90 (MM)
DEDENDUM OF PINION	6.25	4.75 (MM)	TOOTHWIDTH	40.0	40.0 (MM)
DEDENDUM OF GEAR	6.25	7.75 (MM)	ANGULAR SPEED (P)	1450	1450 (RPM)
PROFILE MODIFICATION	NO	YES	TORQUE	275	275 (N-M)

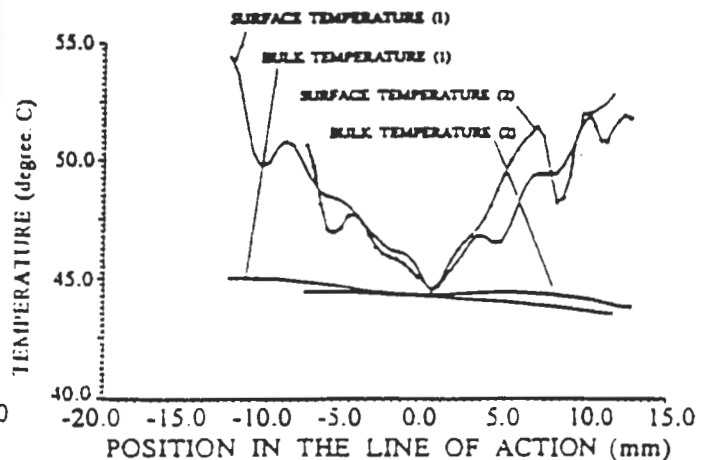
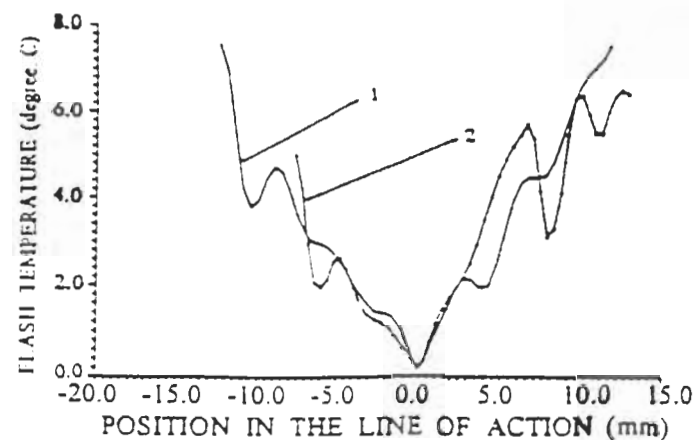
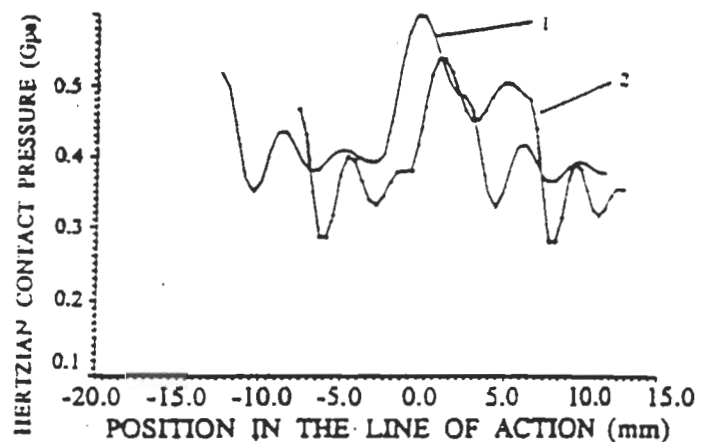
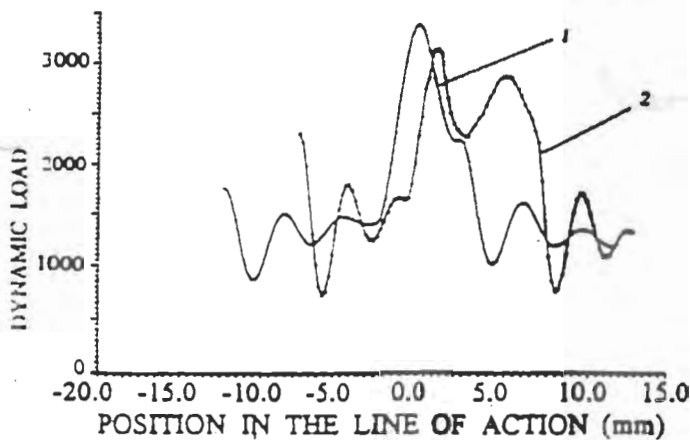
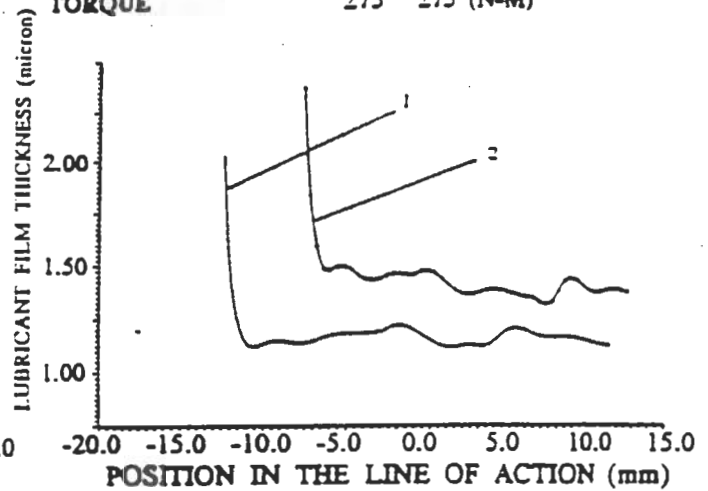
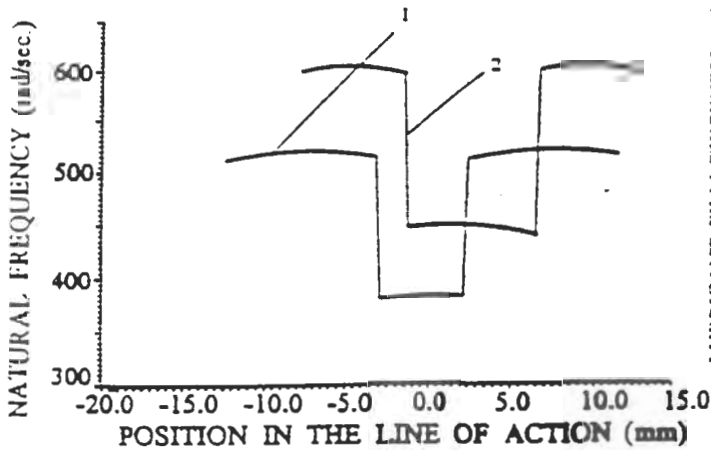


Fig. 12 Comparisons of the Tribological Performances between Standard and Non-Standard Gears

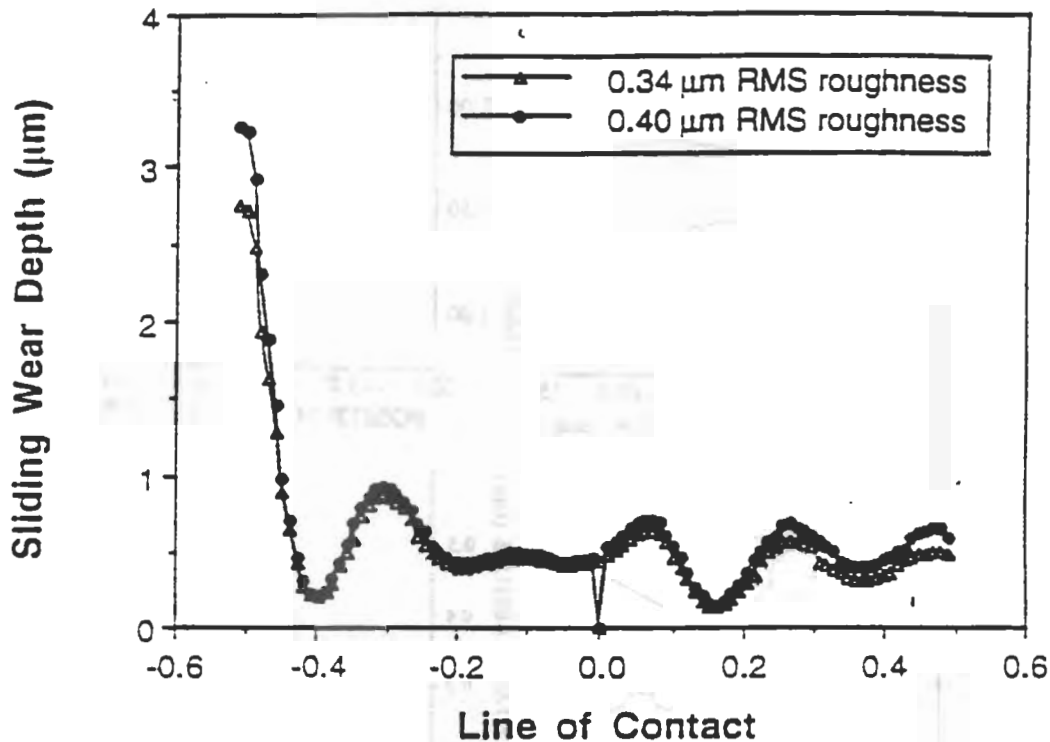
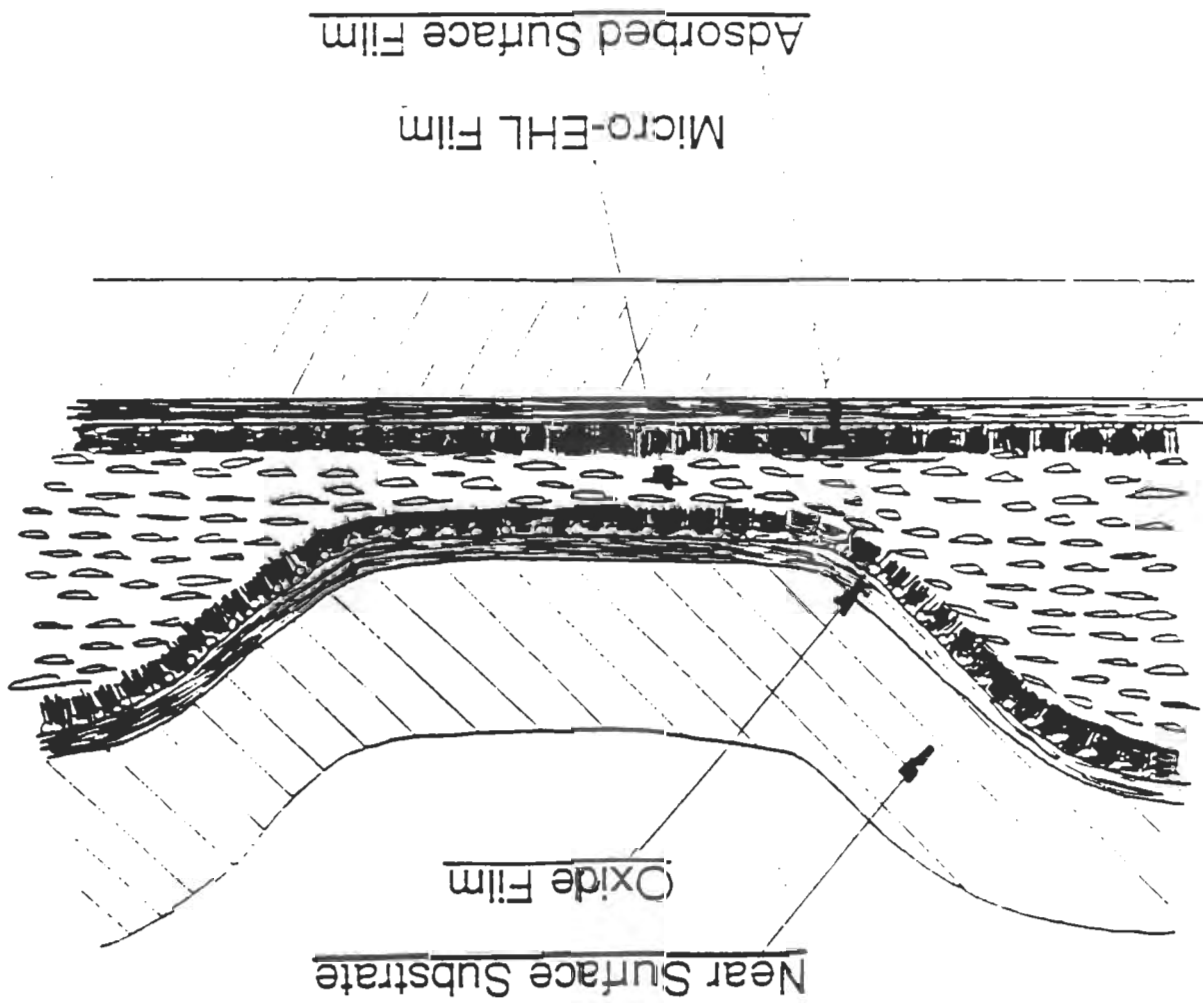
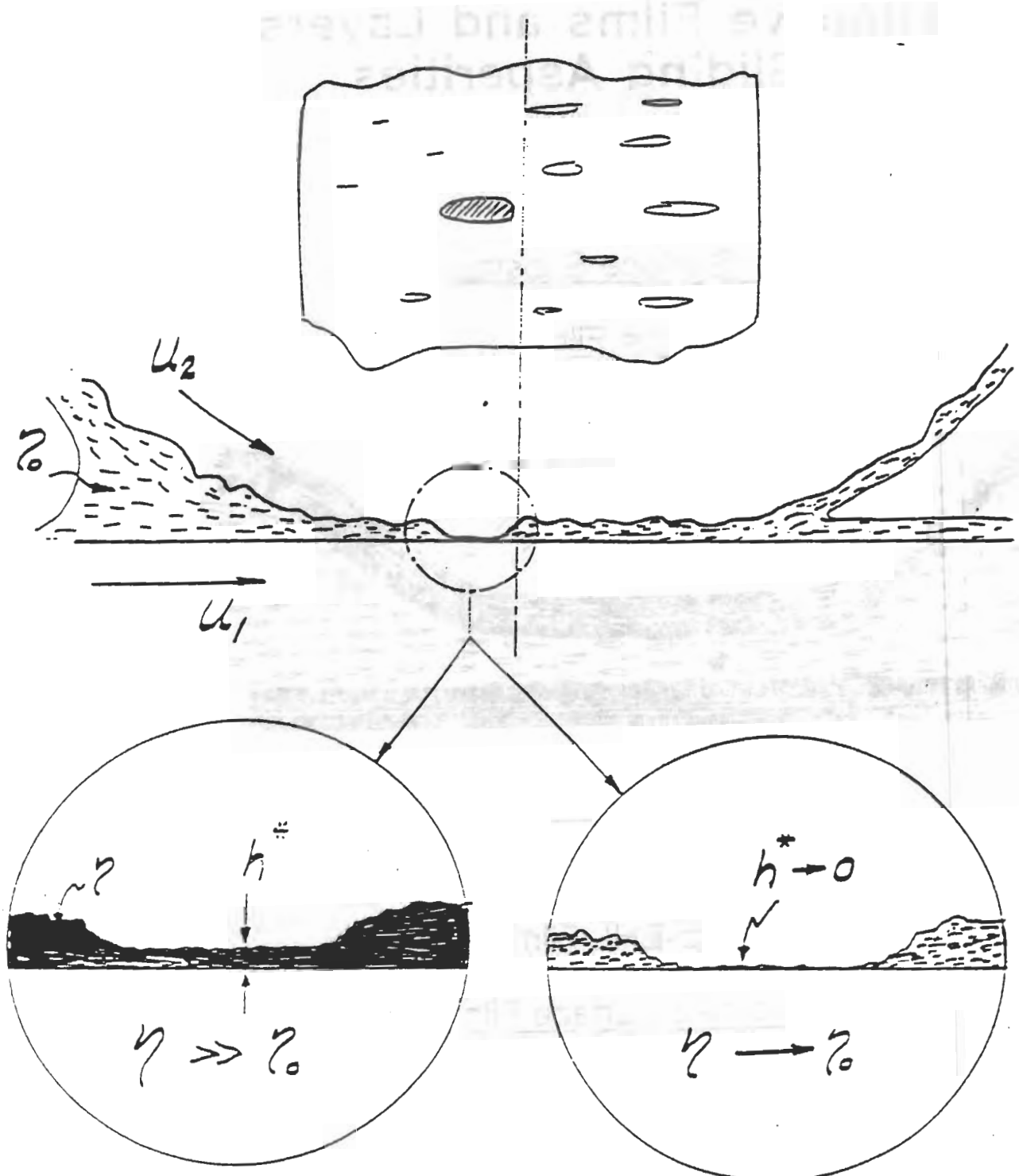


Figure 10. Wear depth distributions or tooth wear profiles after one million cycles

Protective Films and Layers for Sliding Asperities



Micro - EHL Film



Criterion for Breakdown of Micro - EHL Film

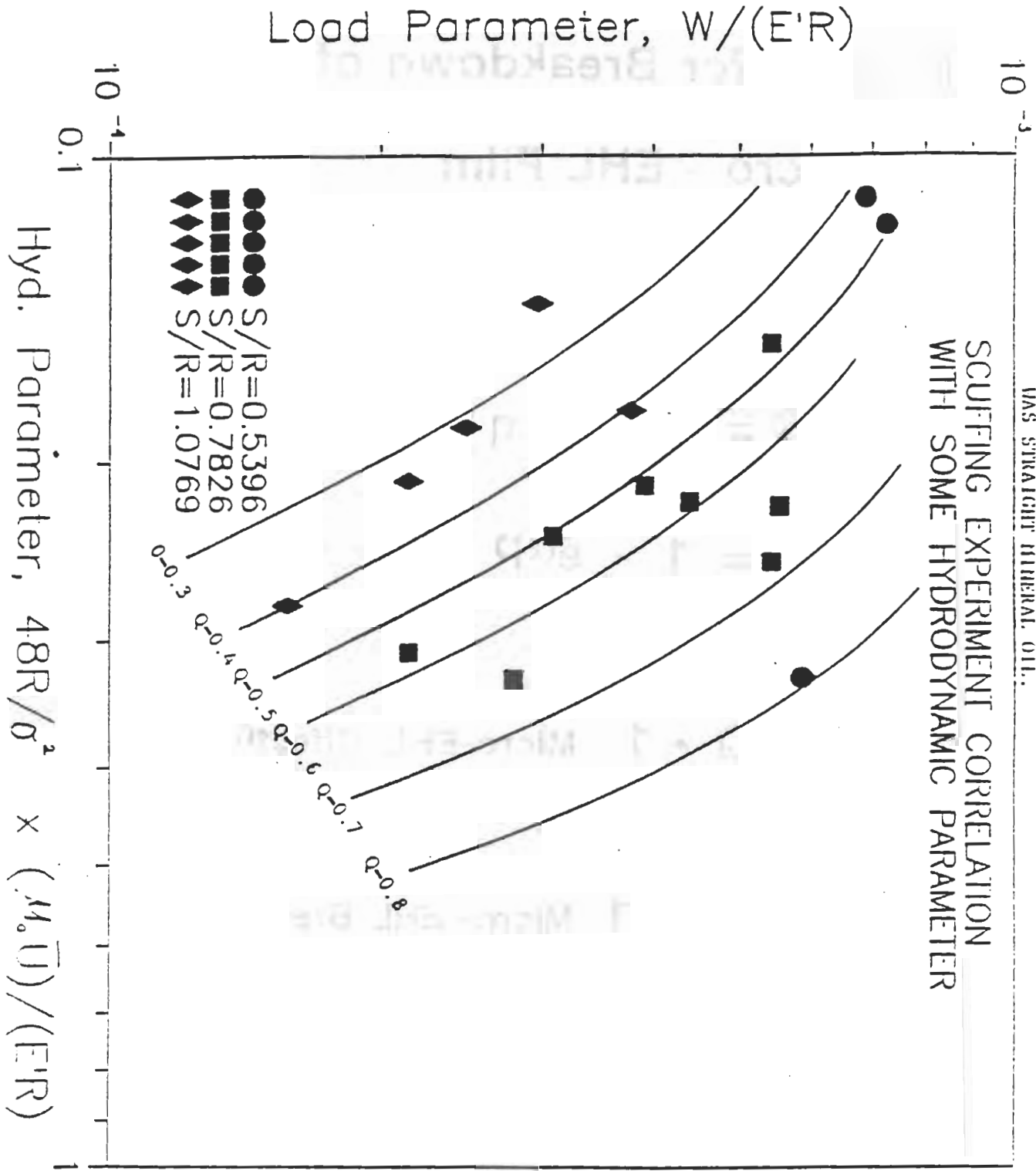
$$Q = 1 - (\eta_0 / \eta)$$
$$= 1 - e^{-\alpha p}$$

$\eta \gg \eta_0$; $Q \rightarrow 1$ Micro-EHL Effective

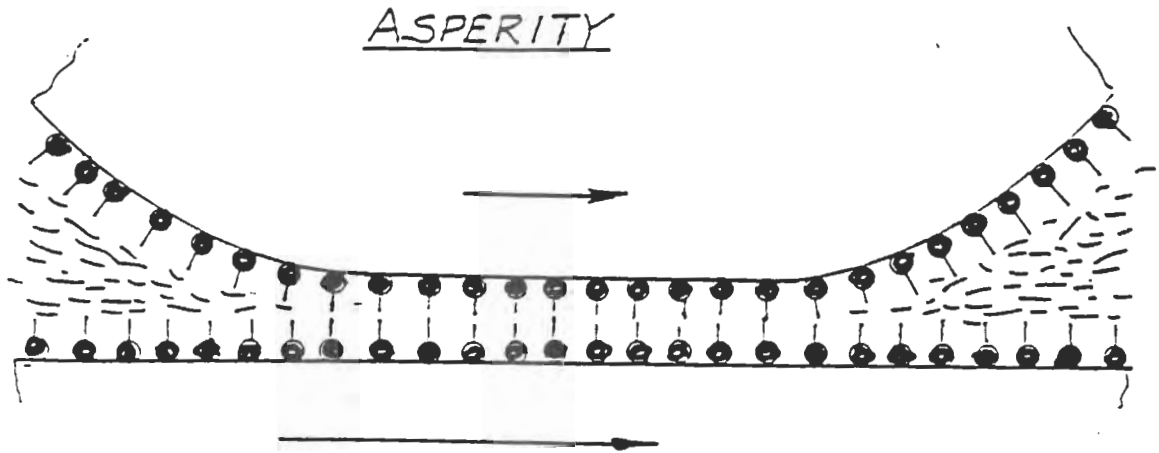
$\eta \rightarrow \eta_0$; $Q < 1$ Micro- EHL Breakdown

SCUFFING EXPERIMENT RESULTS CORRELATED WITH DYSON'S
 Q-MECHANISM THEORY. THIS IS FOR TEST CASES WHICH ORDER
 WENT MACROSCOPIC SCUFFING FAILURE. THE LUBRICANT USED
 WAS STRAIGHT MINERAL OIL.

SCUFFING EXPERIMENT CORRELATION
 WITH SOME HYDRODYNAMIC PARAMETER



ADSORBED SURFACE FILM



ASSUMPTIONS ON LANGMUIR'S EQUATION

- 1) Adsorption is terminated at one monolayer of the adsorbate.
- 2) The molecules are non-interacting but only competes for a fixed number of sites all having the same activation energy E_a .
- 3) The vibrational and rotational excitations of the molecules are negligible.
- 4) The lubricant and the surface is in thermal equilibrium.
- 5) The molecular kinetics for the lubricant are governed by the ideal gas law.

Thus applying Langmuir's adsorption isotherm theory, the fractional coverage of the adsorbate is given by:

$$\theta = \frac{F}{F + \frac{K_b T}{S} \frac{2\pi m K_b T}{S^2} \exp\left(\frac{-\Delta H_{ads}}{K_b T}\right)}$$

where $F = \frac{P}{\sqrt{2\pi m K_b T}}$

this term represents the molecular bombardment rate of the lubricant molecules on to the surface.

MODELING

OF

ADSORBED SURFACE FILM

ADSORPTION AS A FUNCTION OF PRESSURE AND TEMPERATURE

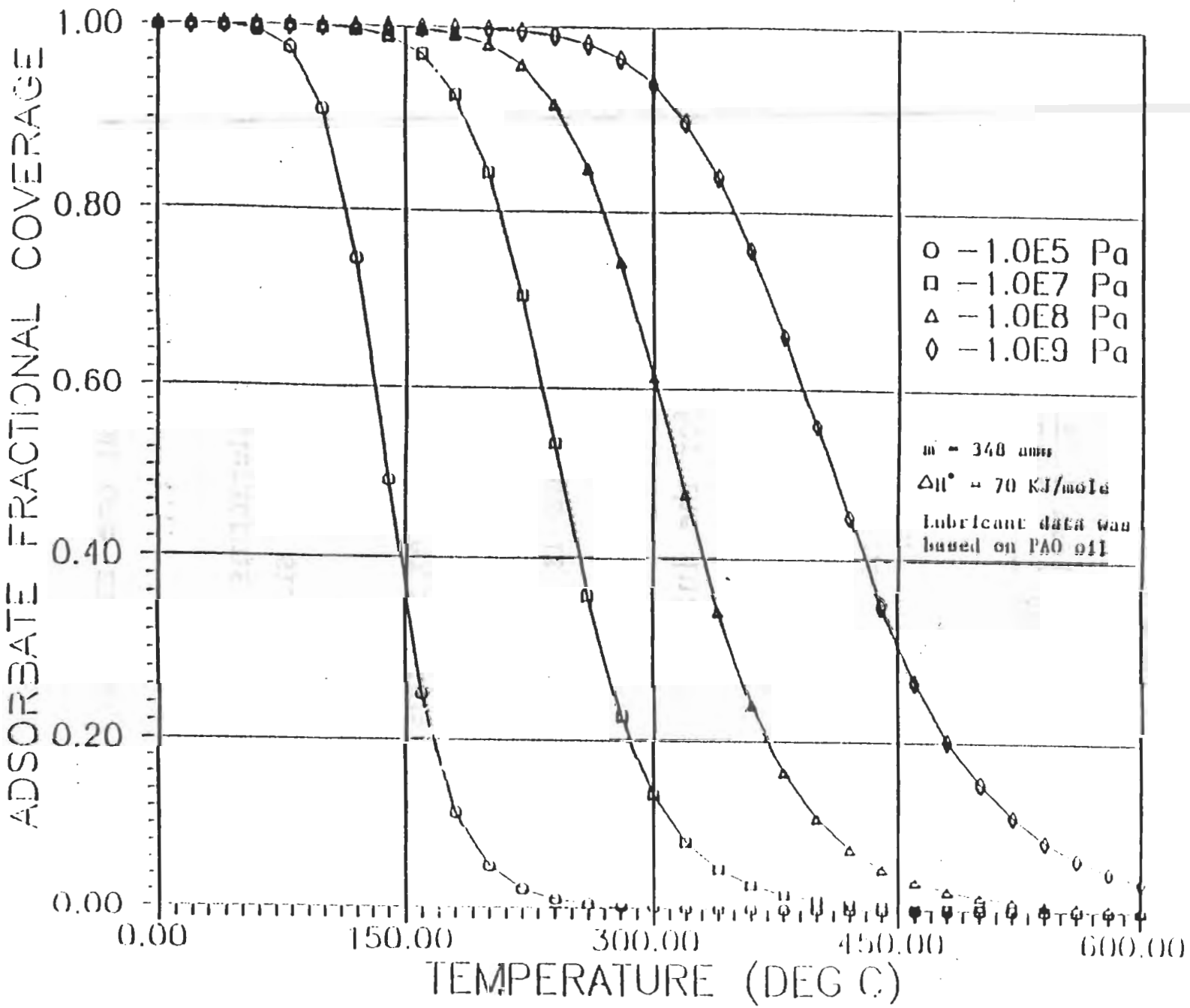
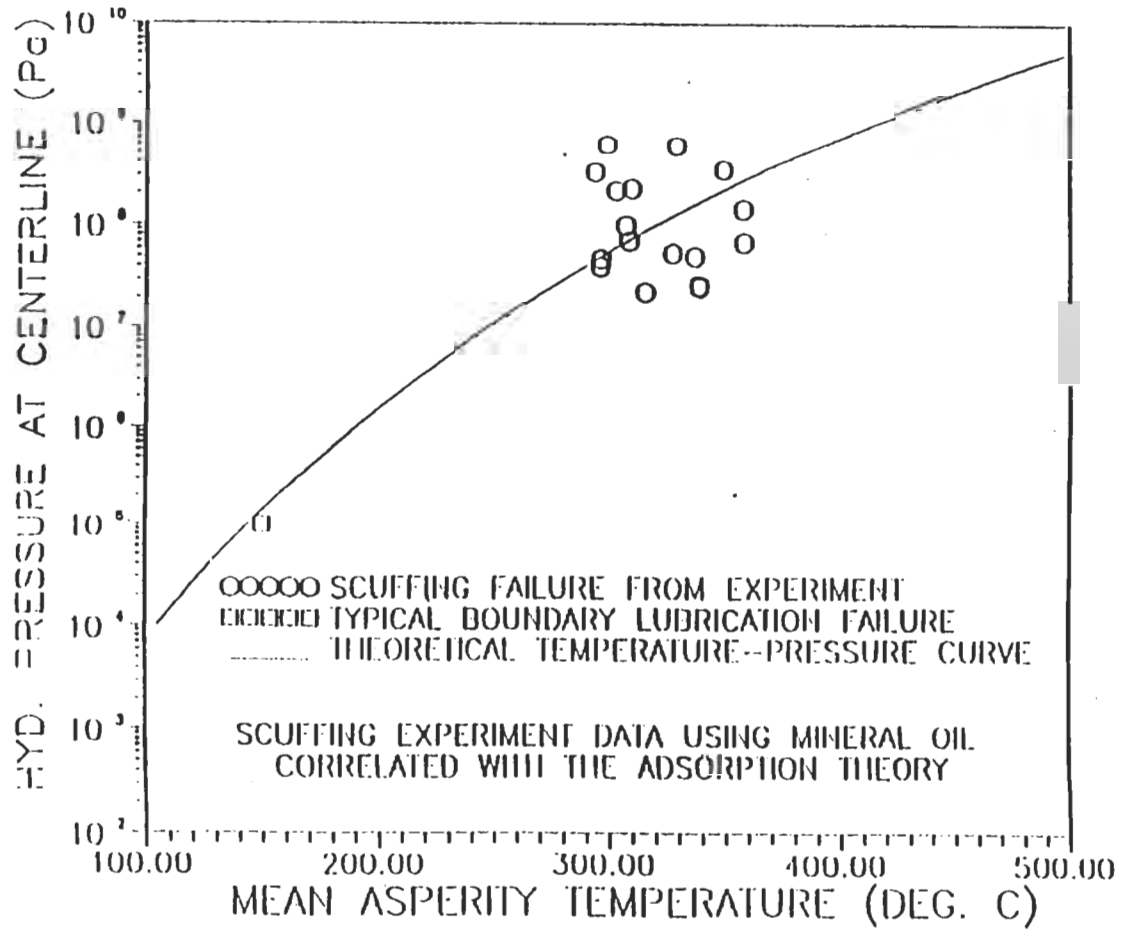
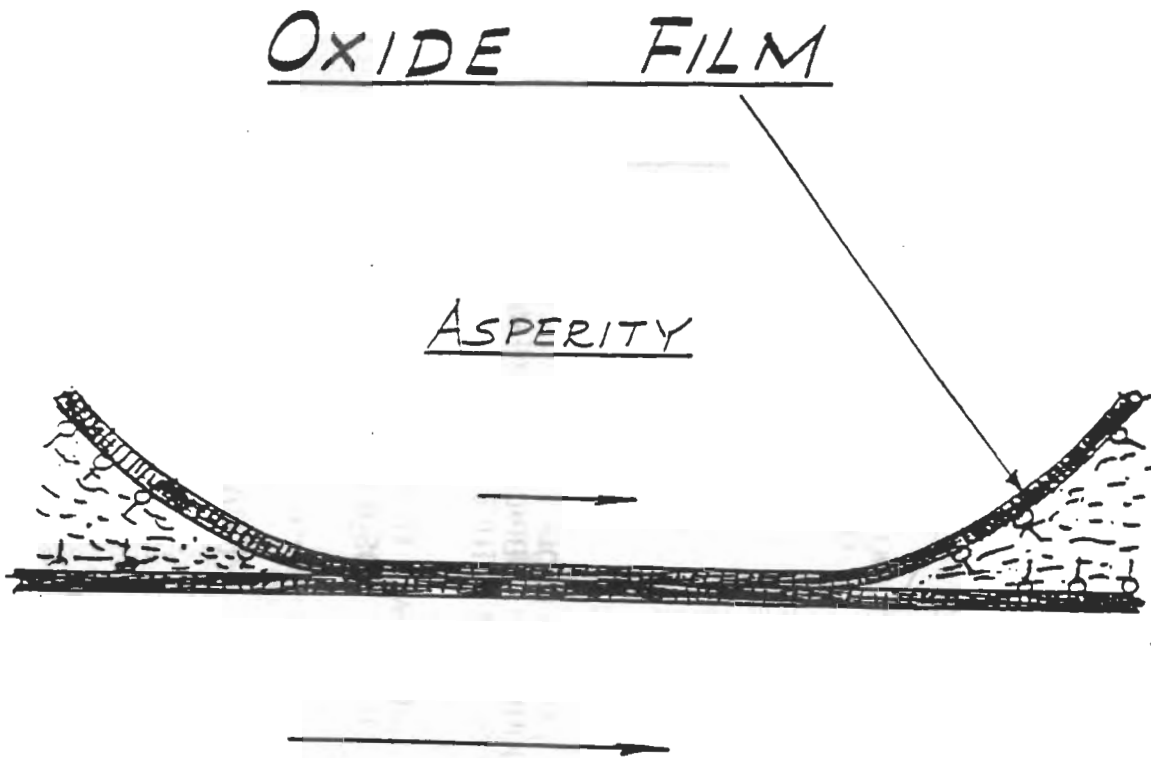


Figure 46





Metal oxide thickness increase:

$$\zeta_m = CA_r e^{\frac{-Q}{RT}}$$

C is a constant dependent upon the metal oxide composition

Metal oxide thickness decrease by Wear:

$$k = k_m \frac{X}{Ut_0} e^{\frac{-E}{RT}} \text{ from C. Rowe's works}$$

$$\zeta_w = k_m \frac{X}{t_0} e^{\frac{-E}{RT}} \frac{Lt}{HA}$$

L = load

H = hardness

A = area

Metal oxidation process:

$$\Delta m = A_r e^{\frac{-Q}{RT}} t$$

A_r = Arrhenius constant

Q = oxidation activation energy

A_r , Q used in the present work are

$$A_r = 7.278 \times 10^{-4} \text{ g/cm}^2 \cdot \text{sec}$$

$$Q = 26 \text{ KJ/mole}$$

Metal oxide thickness increase:

$$\xi_m = CA_r e^{\frac{-Q}{RT}} t$$

C is a constant dependent upon the metal oxide composition

Metal oxide thickness decrease by Wear:

$$k = k_m \frac{X}{Ut_0} e^{\frac{-E}{RT}} \text{ from C. Rowe's works}$$

$$\xi_w = k_m \frac{X}{t_0} e^{\frac{-E}{RT}} \frac{Lt}{HA}$$

L = load

H = hardness

A = area

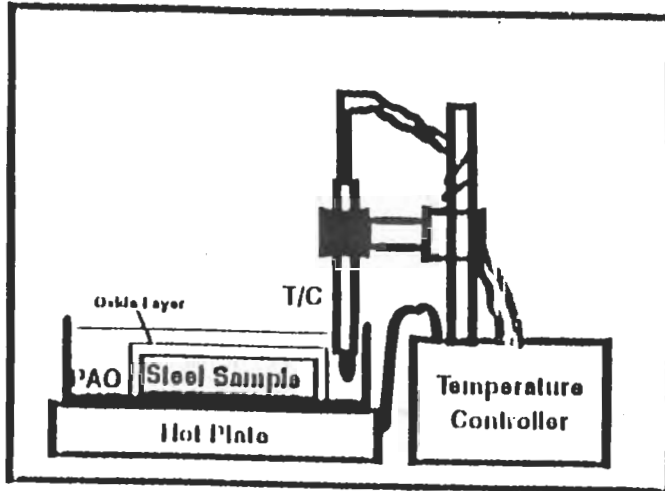
Criterion for scuffing failure to occur:

$$\Delta\xi = \xi_m - \xi_w$$

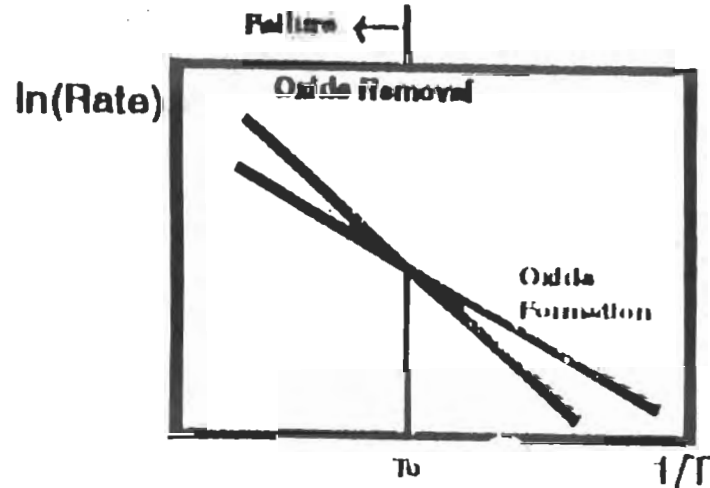
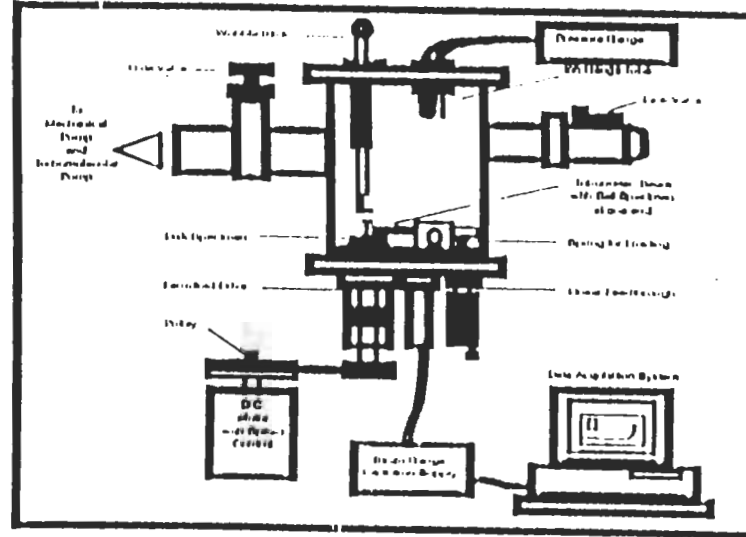
For different operating conditions, when the formation of metal oxide is no longer faster than the removal process by wear, scuffing will occur eventually.

According to the previously mentioned processes, the oxygen content in the lubricant may be totally depleted, which controls the formation and removal of the metal oxide.

Oxidation Test



Oxide Removal in Vacuum

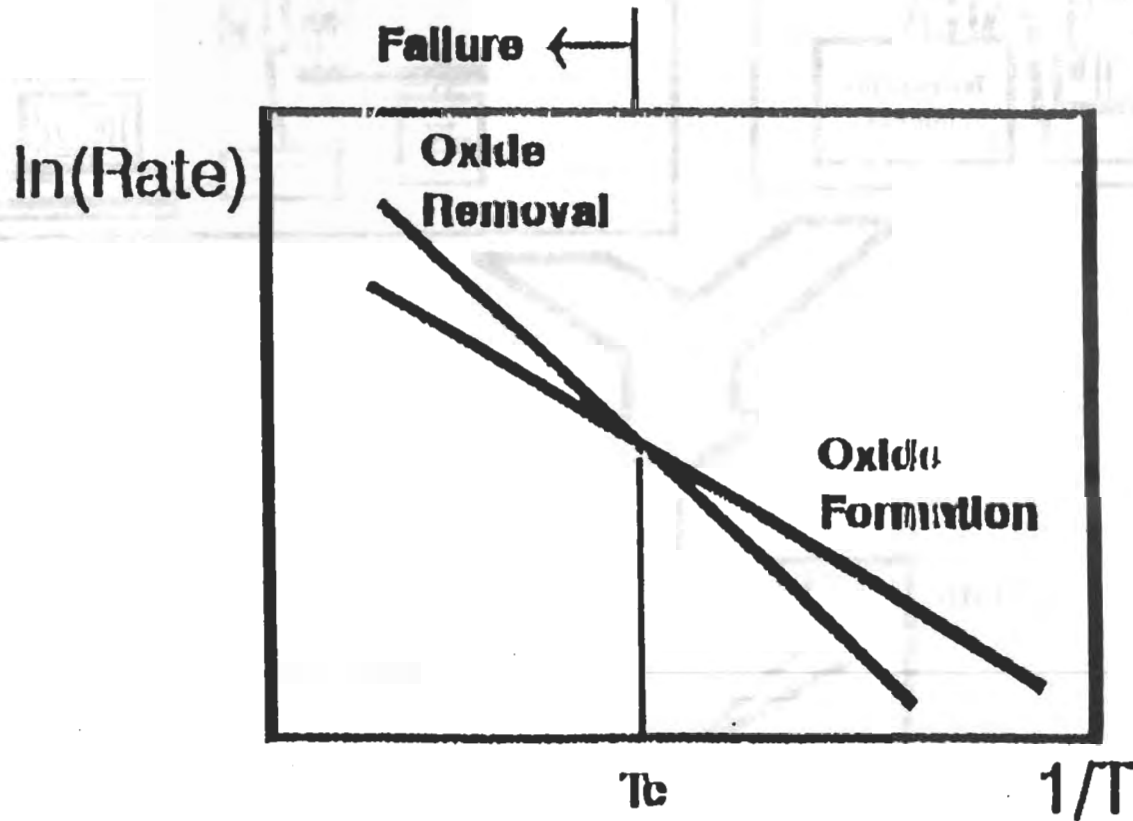


Oxide Formation:

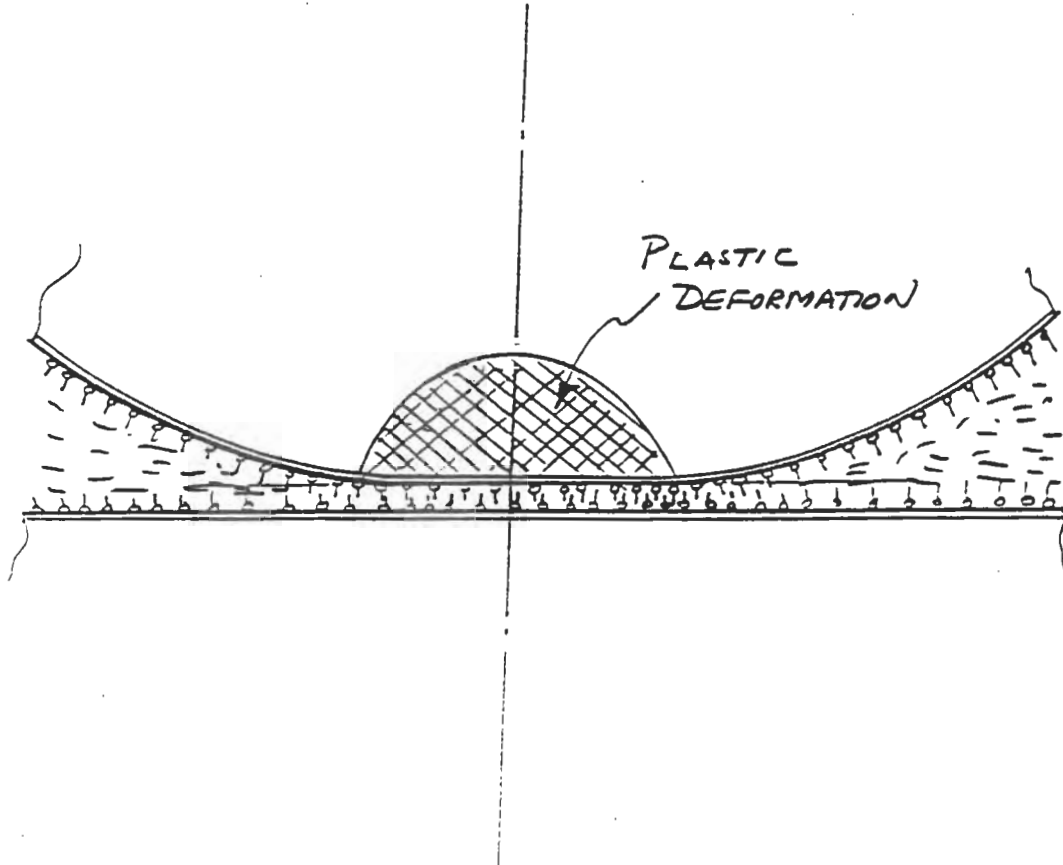
$$R_f = v_f e^{-E_f/RT}$$

Oxide Removal:

$$R_r = v_r e^{-E_r/RT}$$



Failure of Substrate



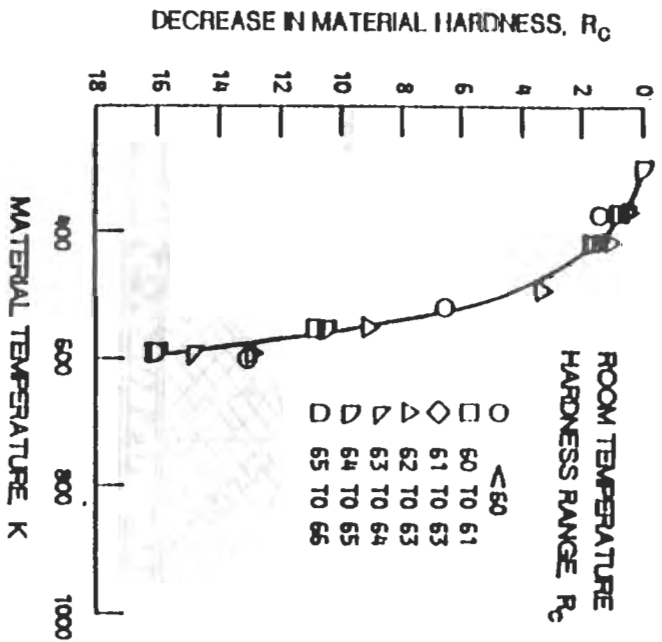


Fig. 7 Hardness of AISI 52100 as a function of Temperature. (From Zaretsky, 1988.)

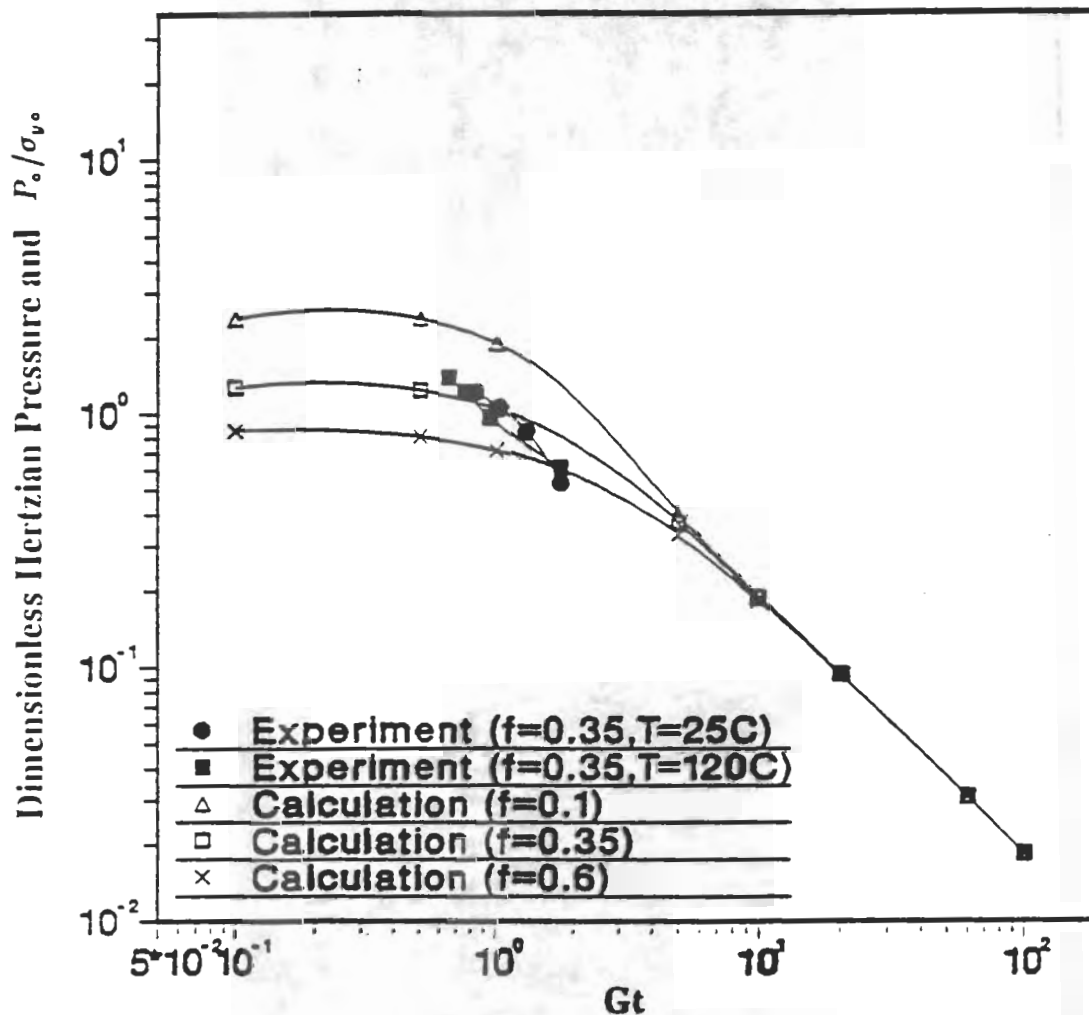
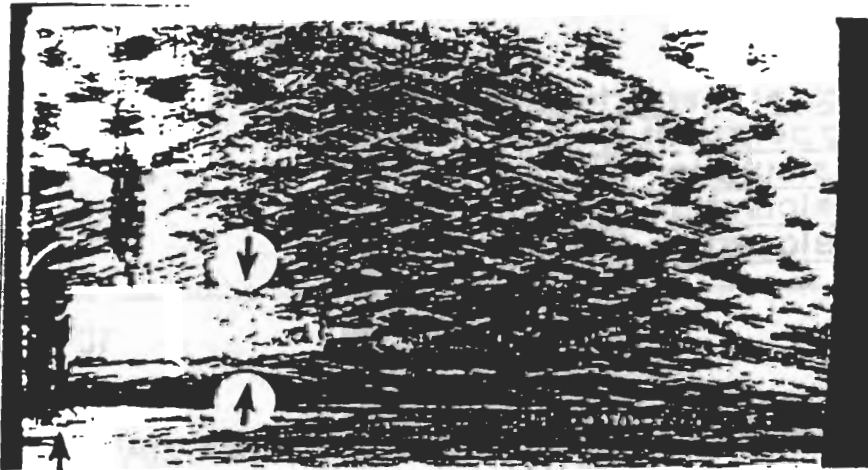
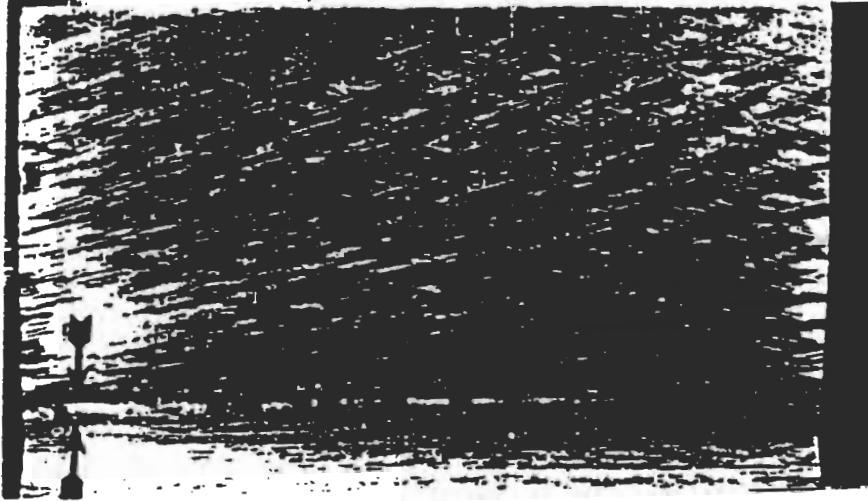


Fig. 9

Comparison between the Calculation and Experimental Data for AISI 52100 in SAE 10W base oil ($Fo=0.1$; $T=25C$ and $T=120C$)



FORM FOR 501

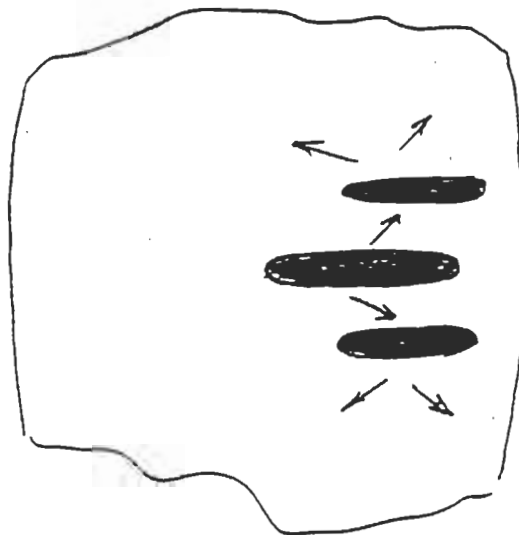
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POSTULATED SCUFFING MECHANISM

- Lubrication Breakdowns Leading to Microscuffing

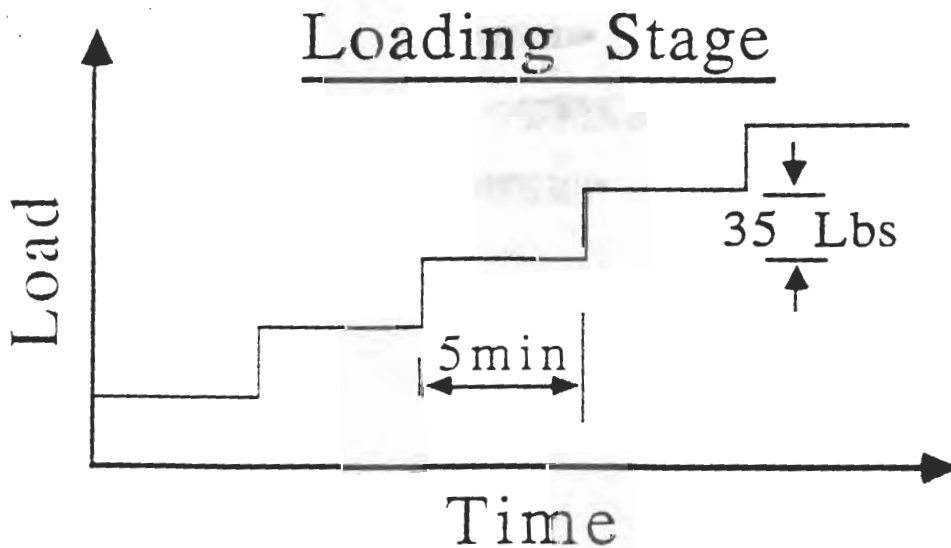
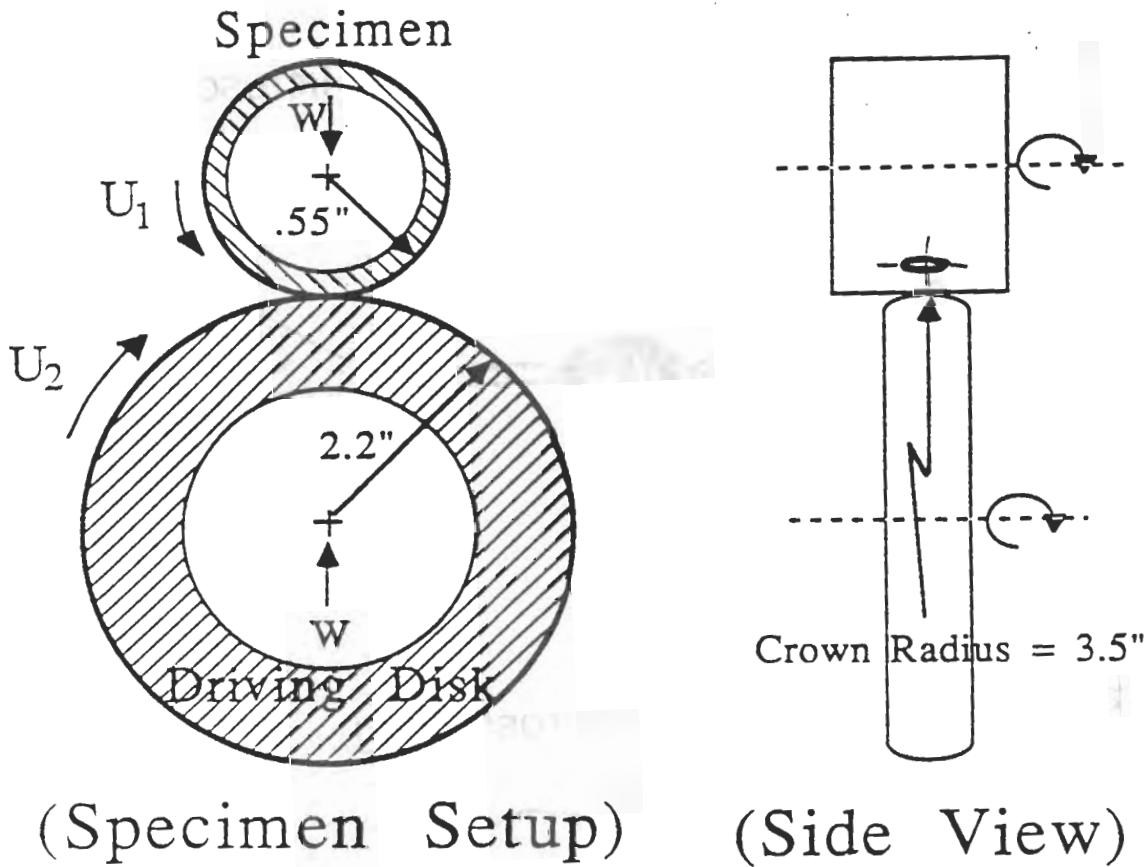


- Transition from Micro to Macroscuffing

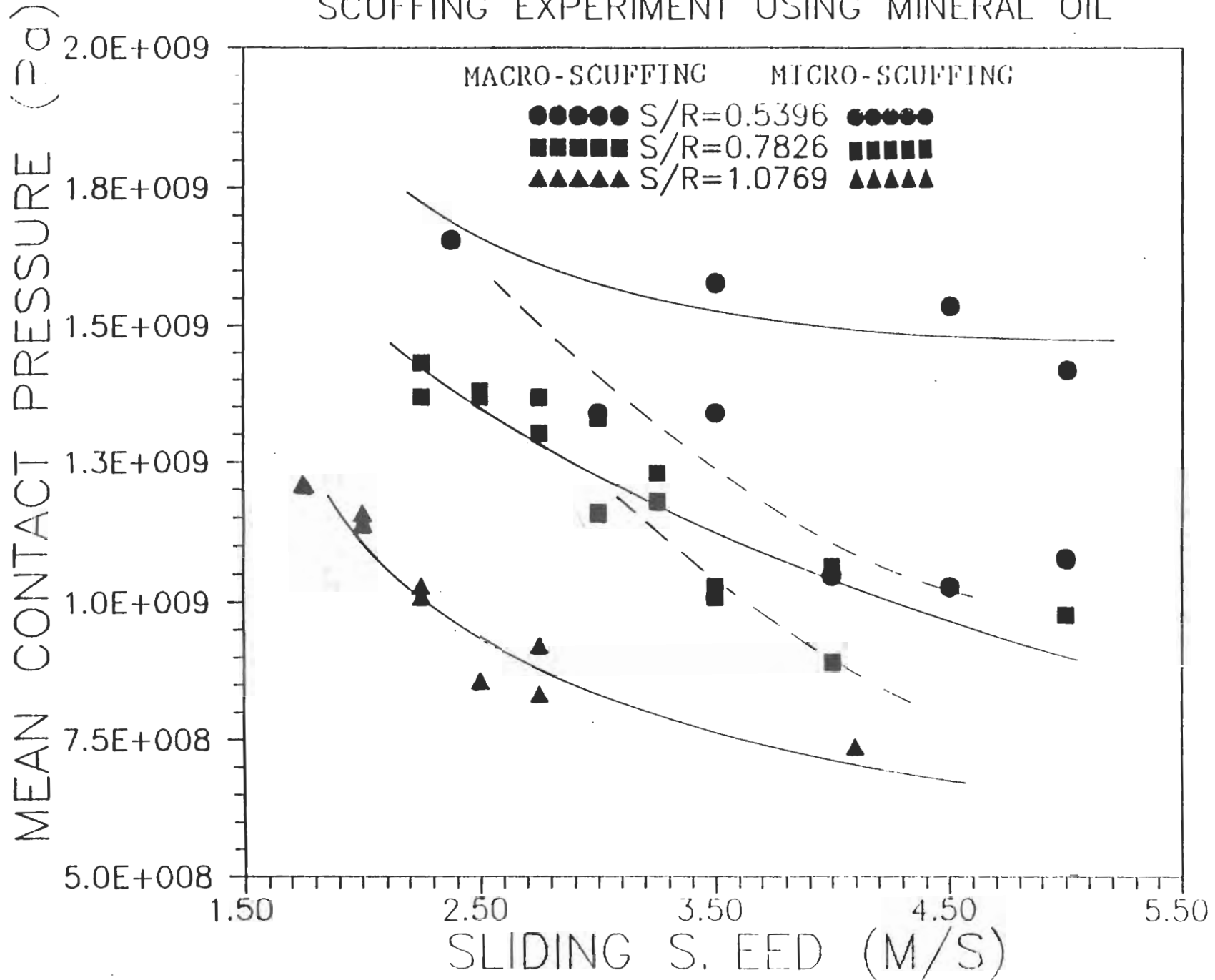


Scuffing Test

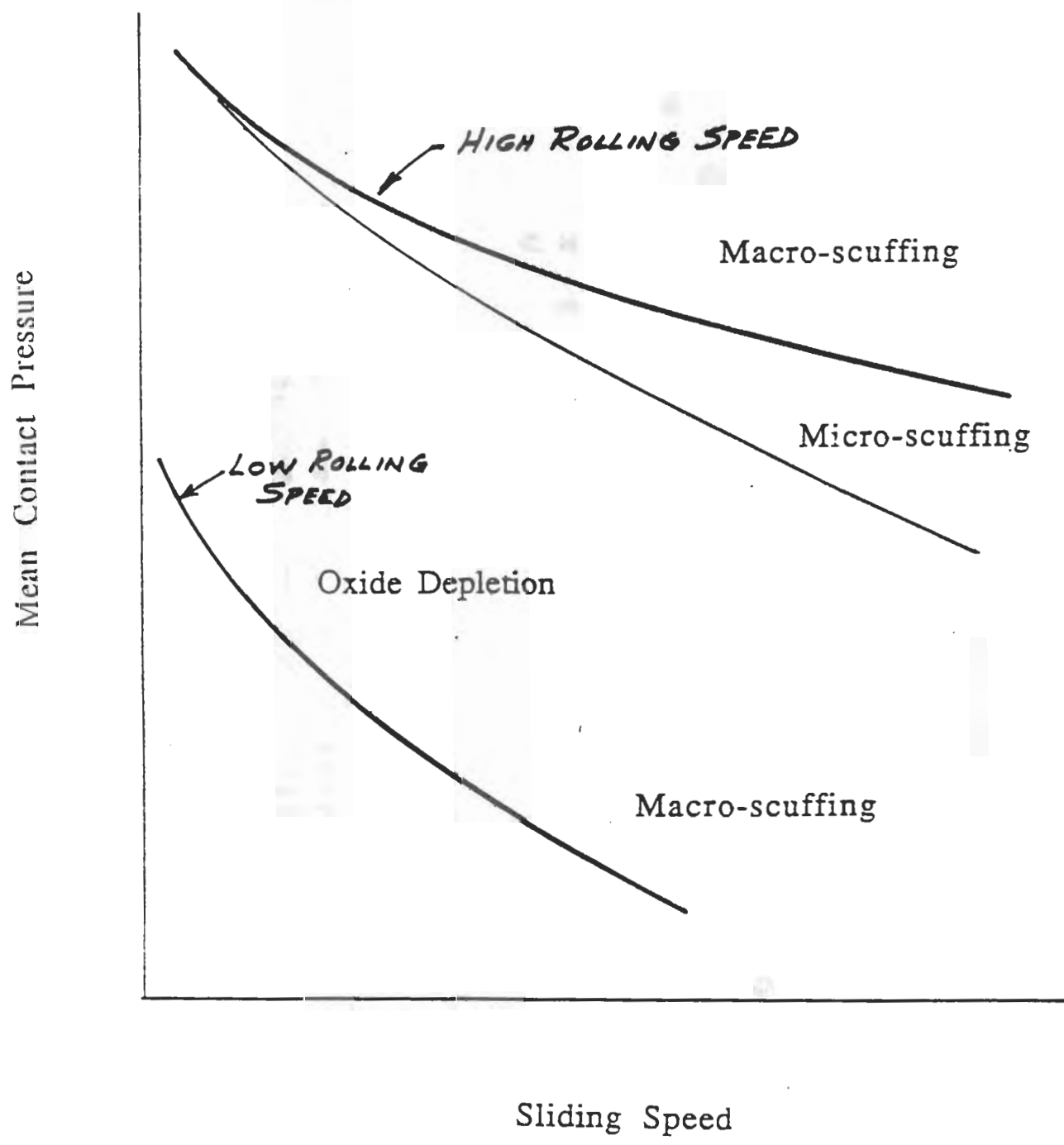
Experimental Setup



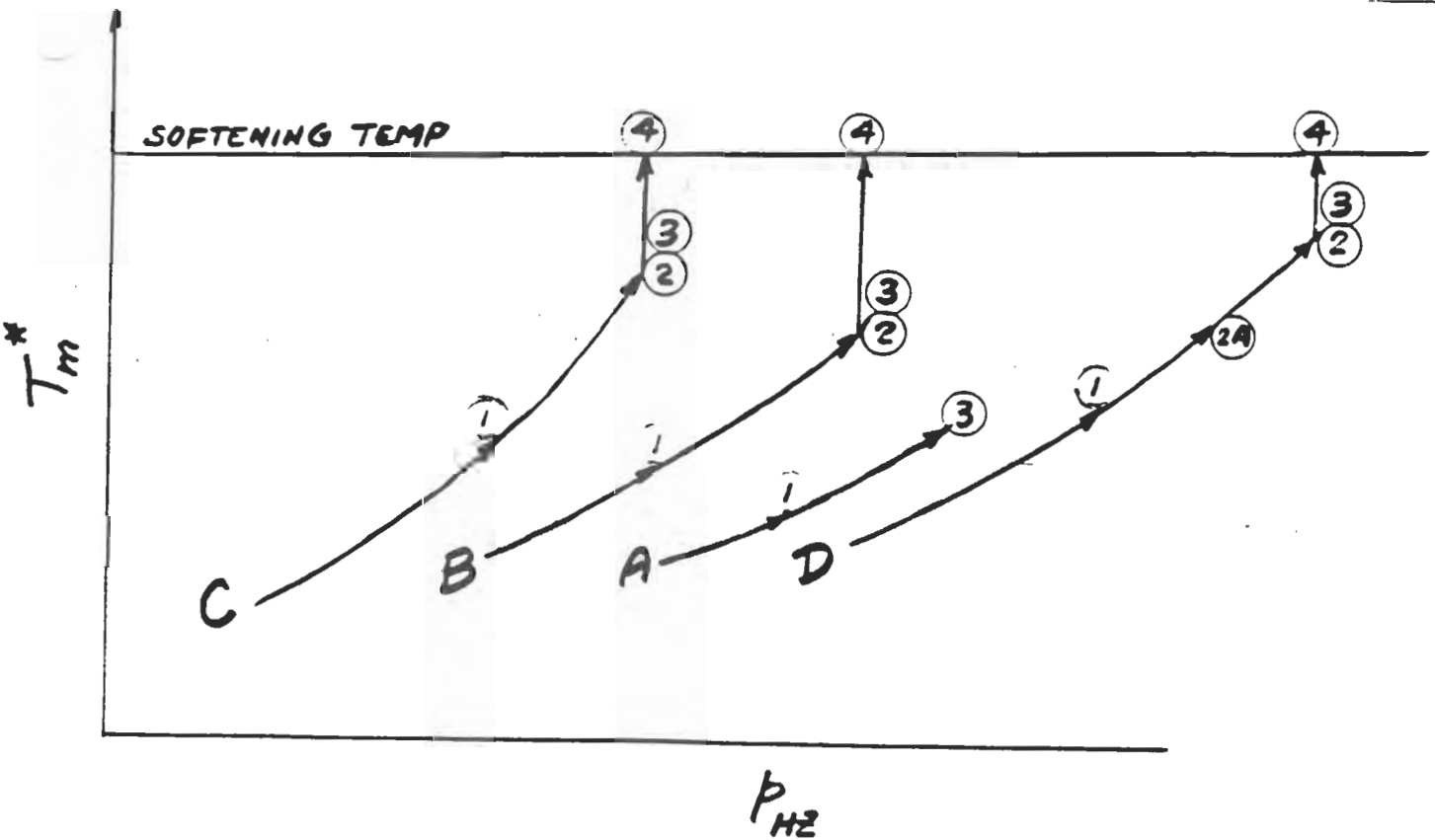
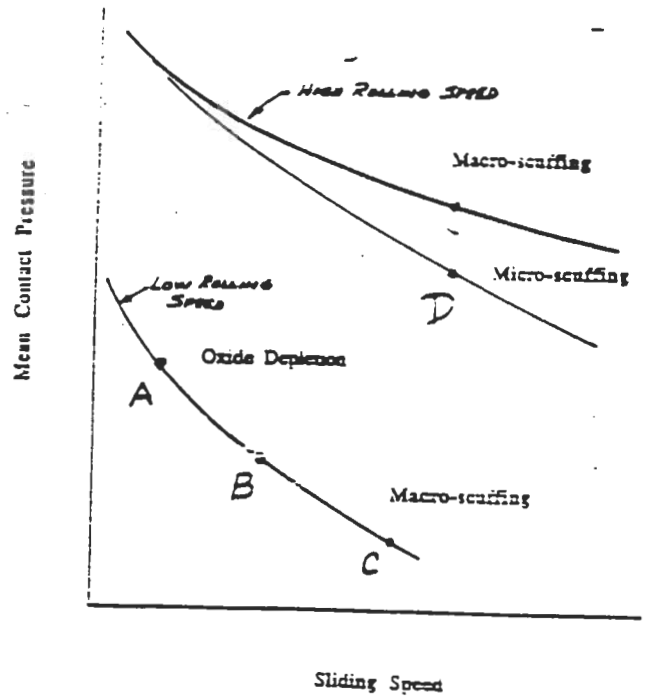
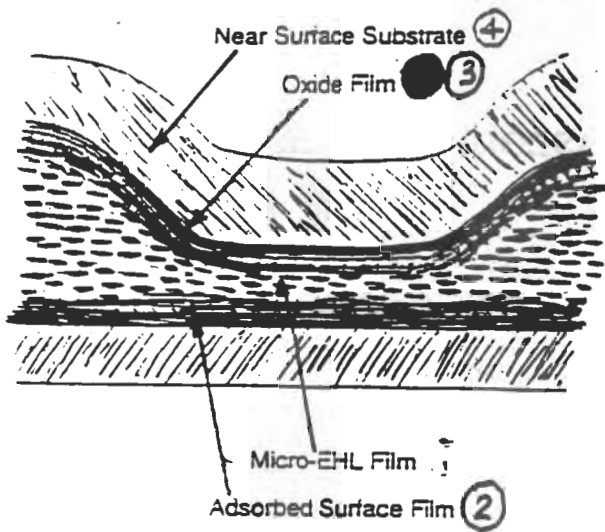
SCUFFING EXPERIMENT USING MINERAL OIL



CONCLUDING REMARKS

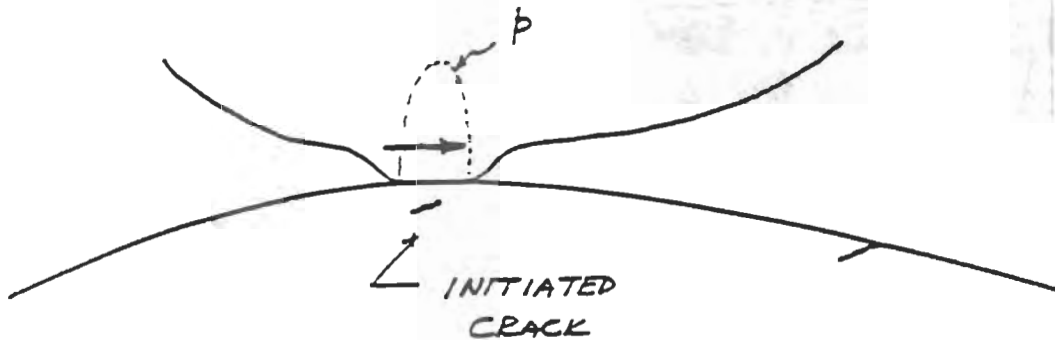


Postulated Successive Modes of Sliding Failure

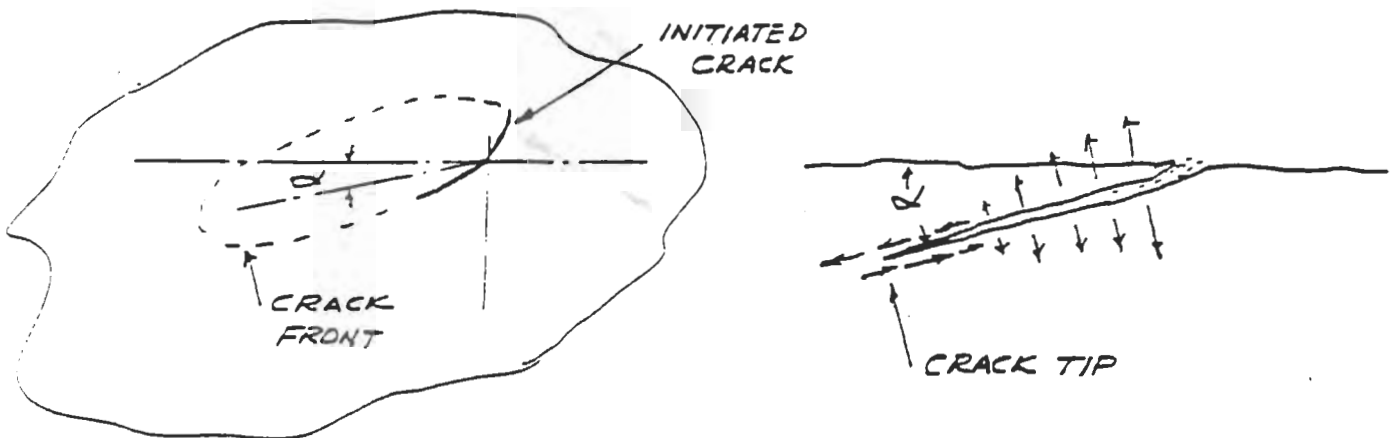


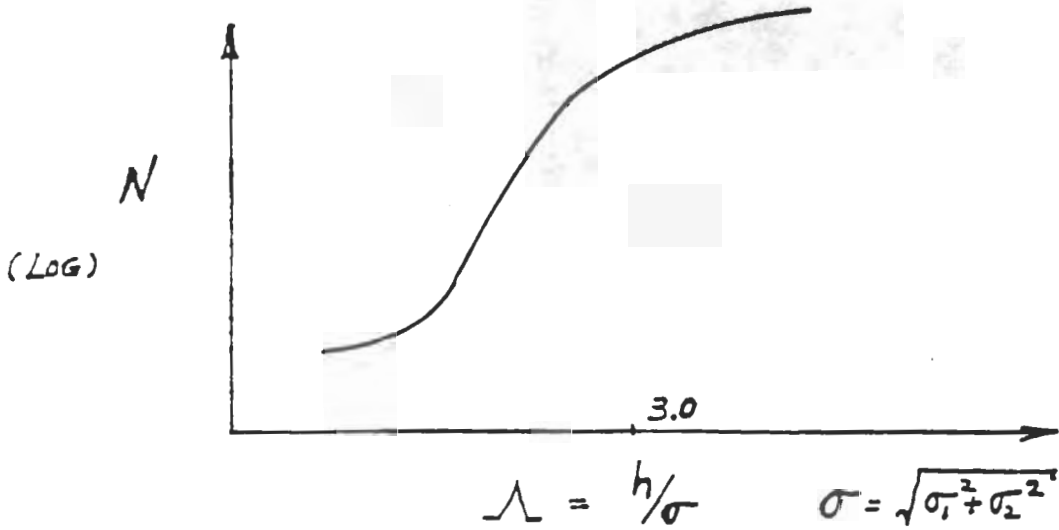
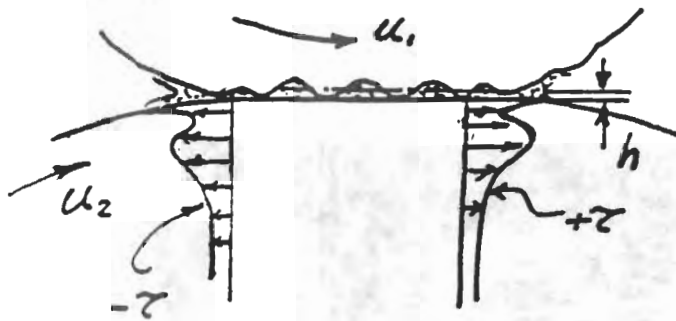
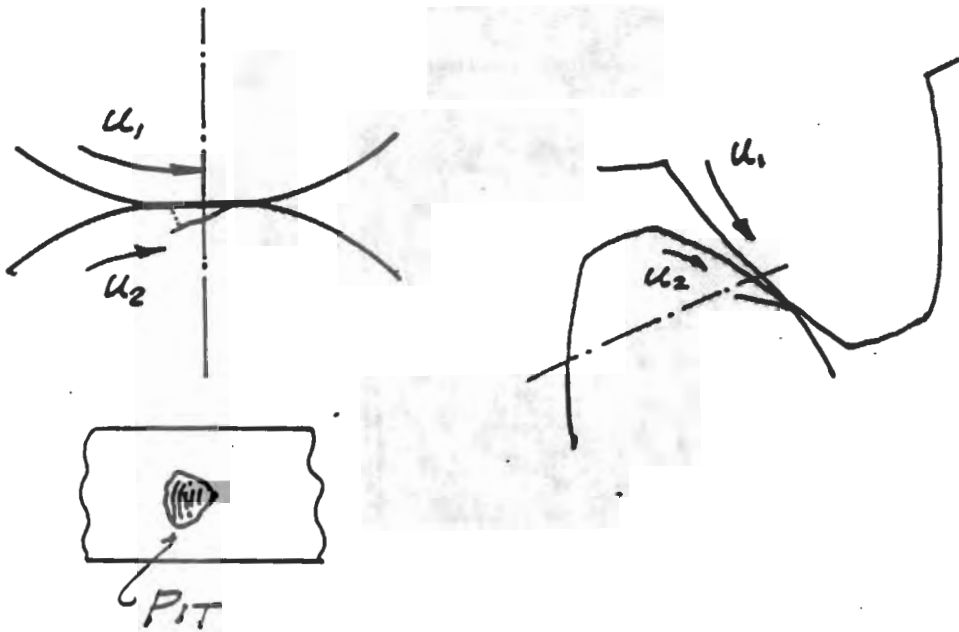
Postulated Mechanism for Surface Pitting

- Crack Initiation from Asperity Contact



- Crack Propagation to Pitting





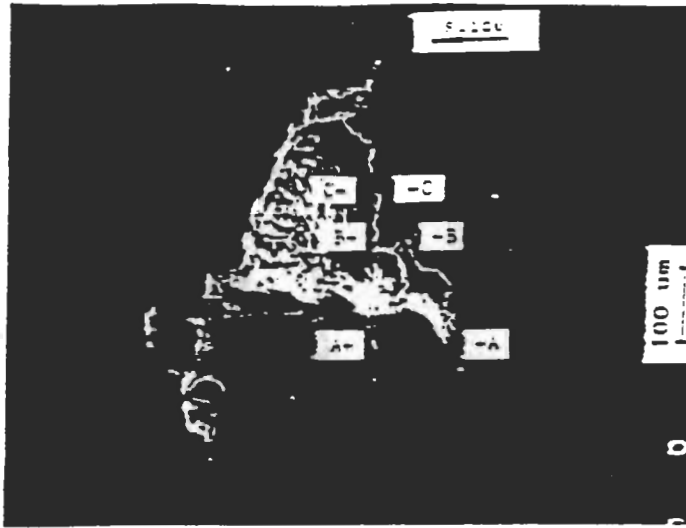


Fig. 3.12 (a) A pit on the surface of S1100 specimen after 4.5 million cycles at 335 KSI maximum Hertzian contact pressure and 0.25 slide-to-roll ratio

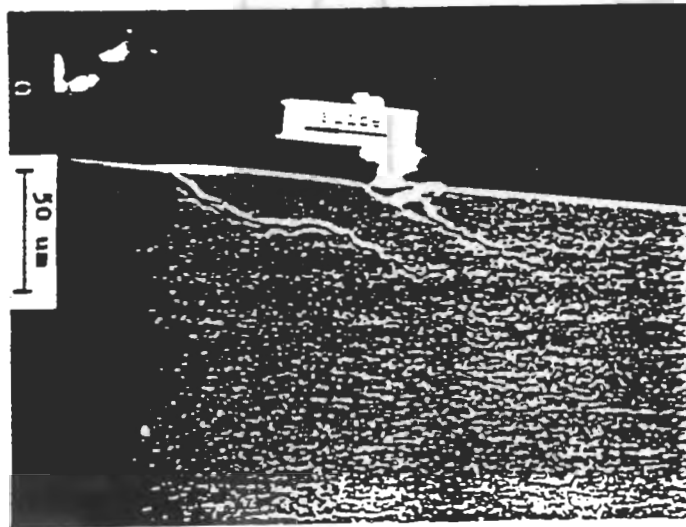
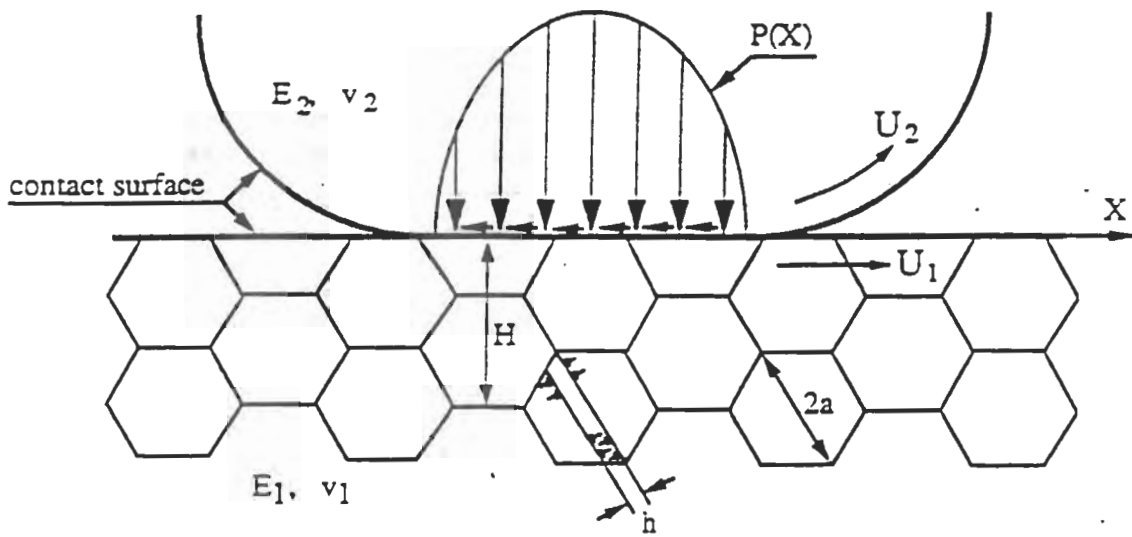


Fig. 3.12 (b) A-A section of the pit in Fig. 3.12 (a).



$2a$ = grain or subgrain size
 h = width of slip band.
 B = half-width of Hertzian contact zone
 H = depth of crack initiation from contact surface

Three cases:

- (i) $0 \leq H < 2 \mu\text{m}$, surface crack initiation;
- (ii) $2 \mu\text{m} < H < 0.4B$, near surface crack initiation;
- (iii) $0.4B < H$, substrate crack initiation.

Fig. 1. Modeling of crack initiation under contact fatigue.

Surface Crack Initiation Life, n_i

Fig. 8. Model prediction of effect of hardness on crack initiation life ($q=3.8$).

Handwritten notes: *Model prediction for crack initiation*

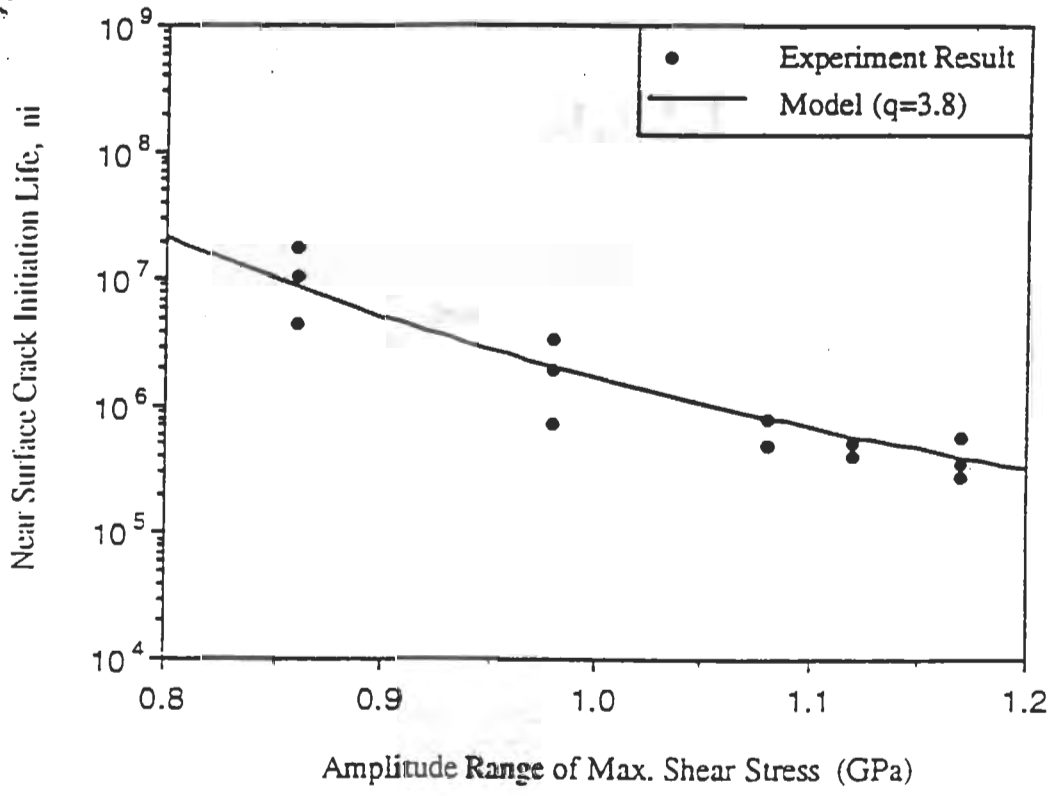


Fig. 7. Comparison of model prediction with experimental results (Zhou, et al. 1989) about near surface crack initiation life ($R_c = 62.5$).

PROBLEM CONFIGURATION FOR CONTACT FATIGUE

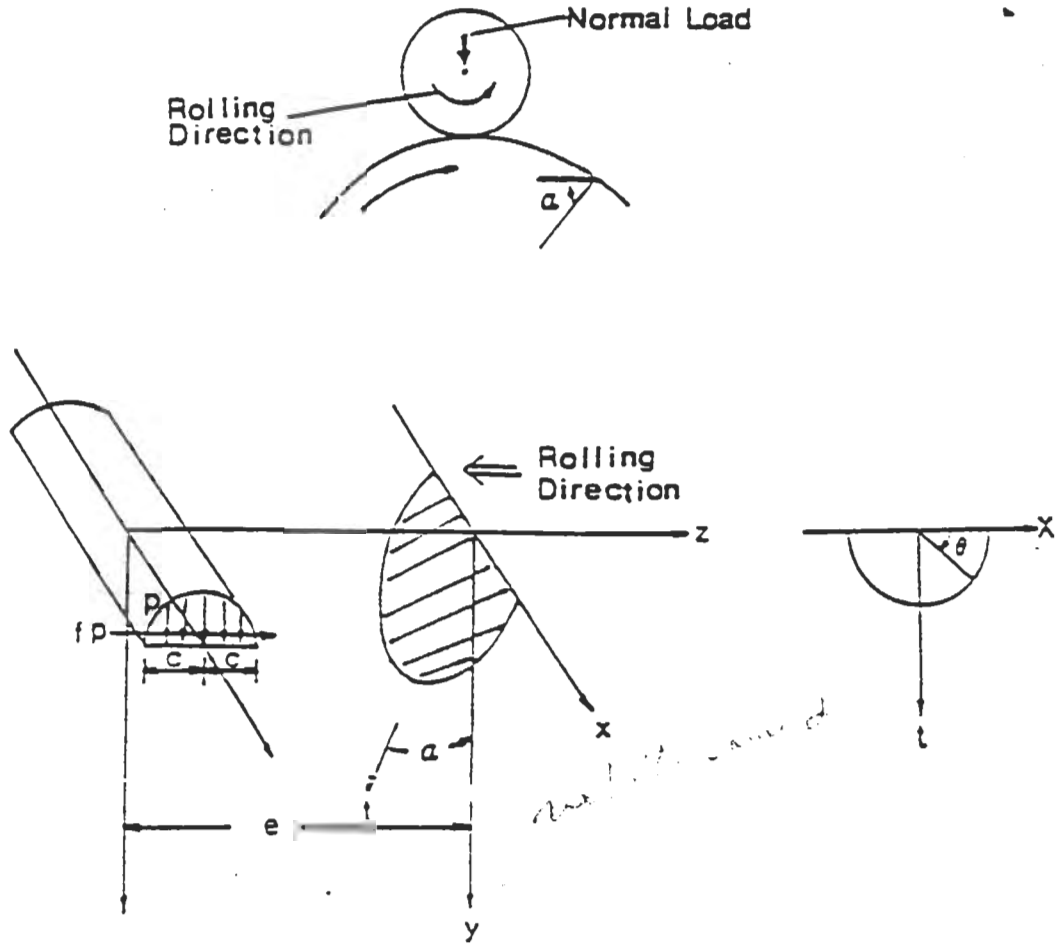


Figure 3. Geometry and coordinate system for crack propagation model.

$$\alpha = -67.5$$

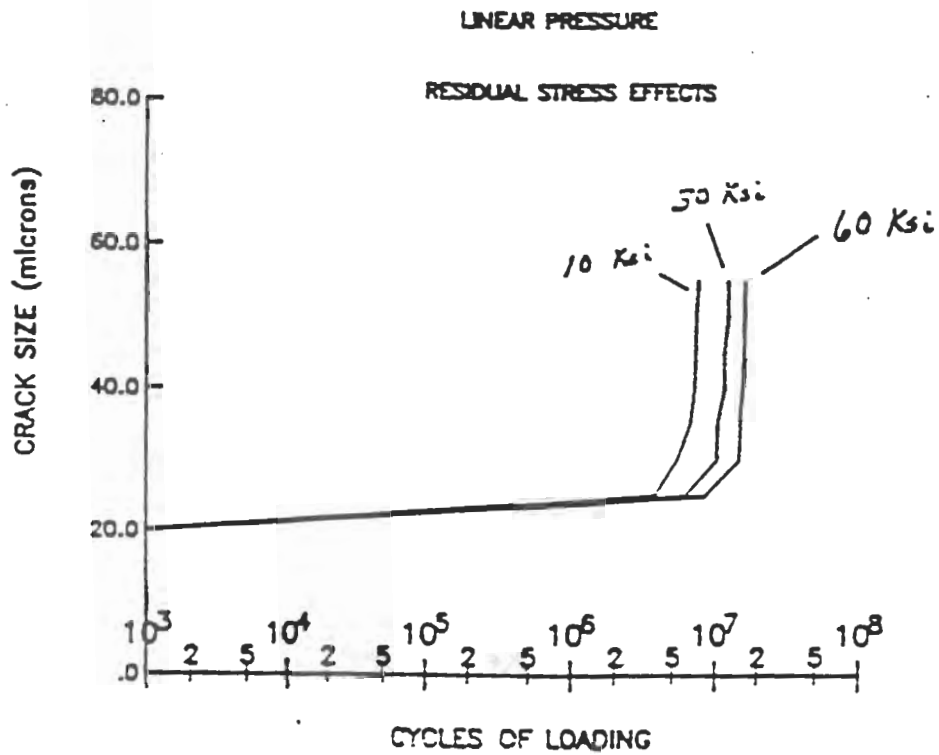


Figure 11 Predicted crack size versus loading cycles for a crack inclined at 22.5 degrees from the surface under linear fluid pressure including residual stress effects.

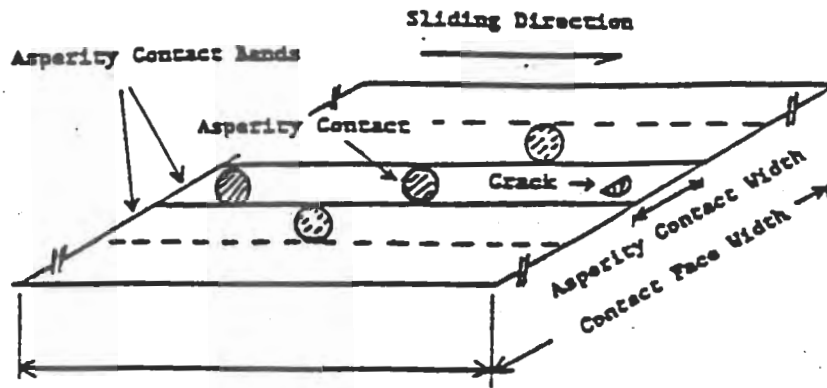


Fig. 4 Contact surface showing asperity contact band areas

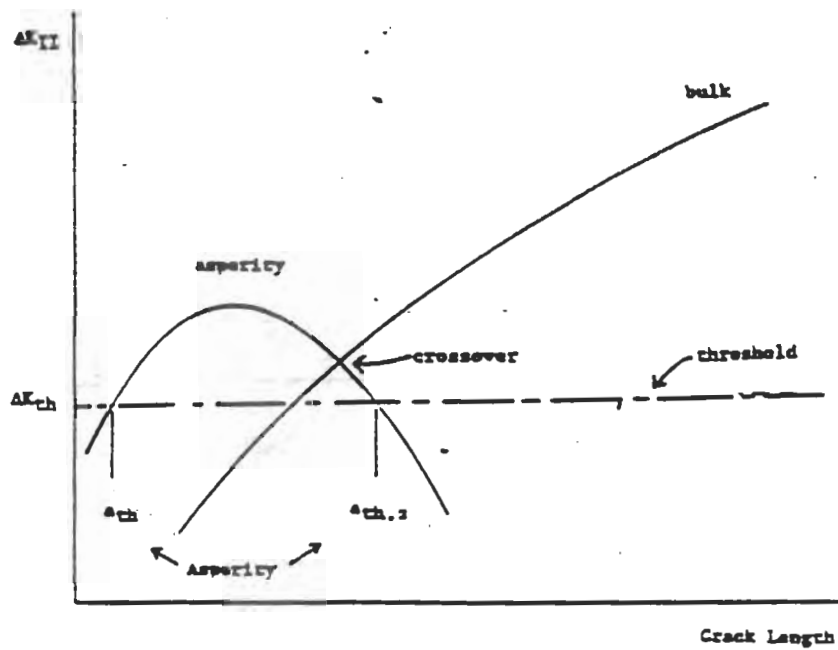


Fig. 3 Crack tip stress intensity ranges induced by asperity and bulk contact loading

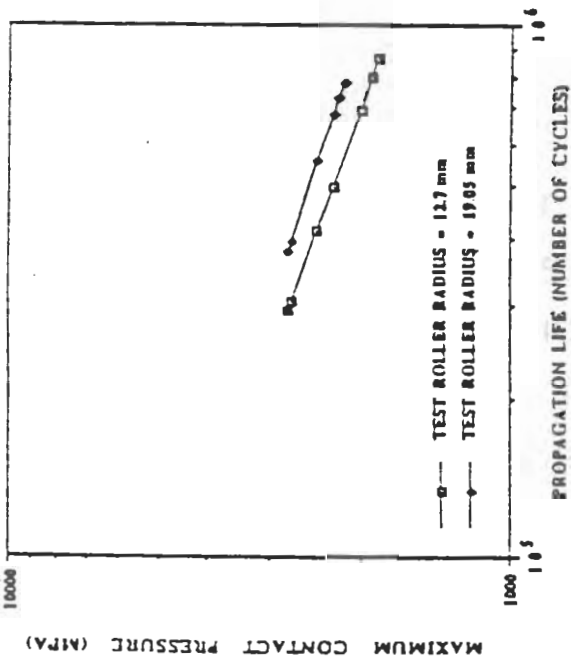


Fig. 5 Effect on predicted pitting life of varying load and contact radii

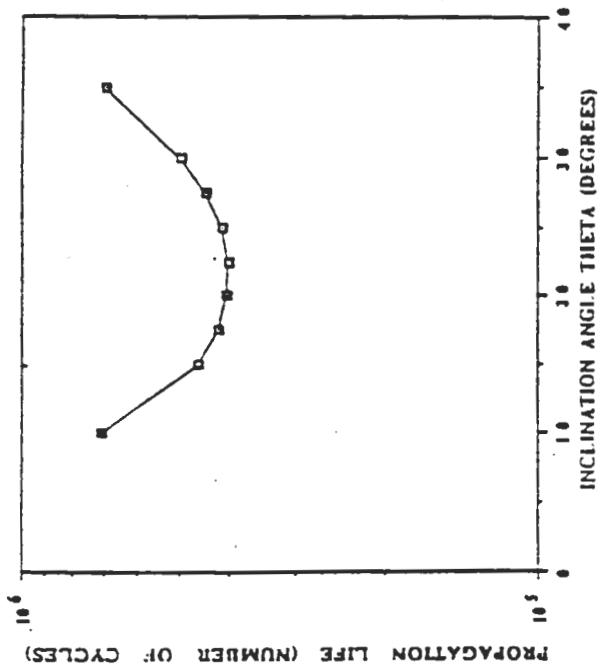


Fig. 6 Effect on predicted pitting life of changes in the crack inclination angle

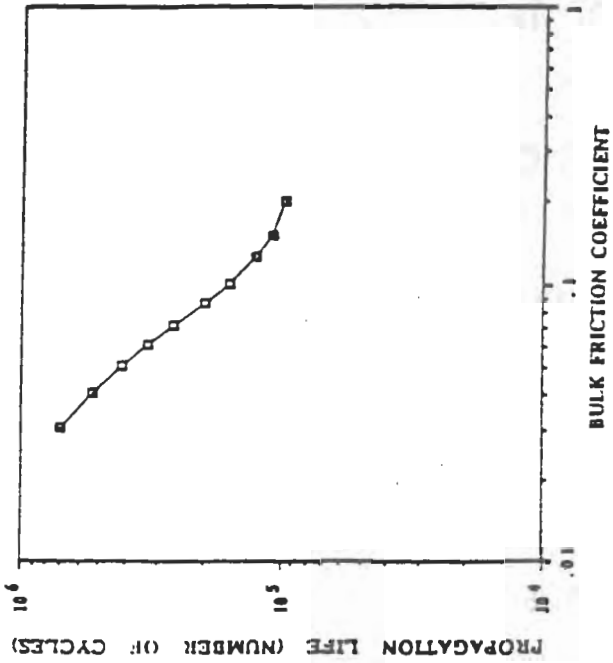


Fig. 7 Change in predicted life with variation in the bulk friction coef. ficient

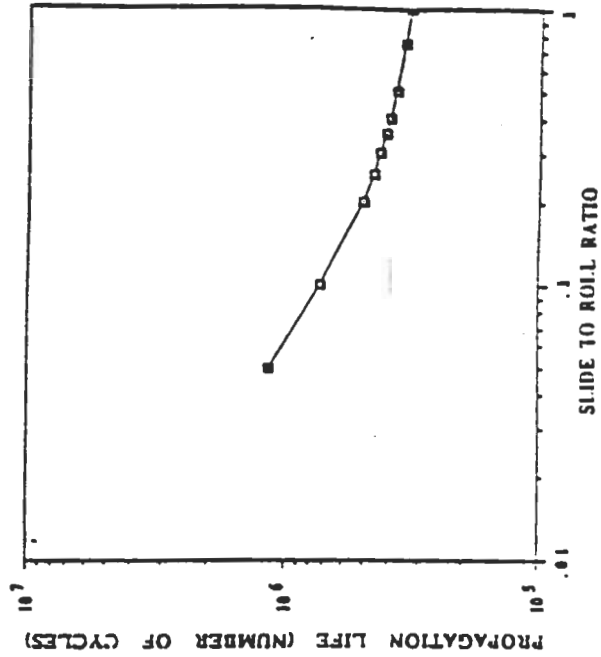


Fig. 8 Effect on predicted life of changes in the slide to roll ratio.

CONCLUDING REMARKS

- Both surface fatigue and scuffing are controlled by mixed lubrication through the average film thickness, asperity pressure, friction, and asperity temperature.
- Existing gear lubrication analyses such as TELSGE for spur gears are useful for improving predictions of surface fatigue and scuffing using current AGMA criteria with the maximum contact pressure and total temperature.
- Recent surface fatigue life model shows merit and promise leading to an improved life prediction based on fracture mechanics.
- Recent scuff experiments and modeling for rollers suggest that scuffing is generated by micro-scuffing of asperities due to lubrication breakdowns leading to macro-scuffing of an extended region, and correlate with the asperity contact temperatures.